

# Na<sub>2</sub>TeS<sub>3</sub>, Na<sub>2</sub>TeSe<sub>3</sub>-*mP24*, and Na<sub>2</sub>TeSe<sub>3</sub>-*mC48*: Crystal Structures and Optical and Electrical Properties of Sodium Chalcogenidotellurates(IV)

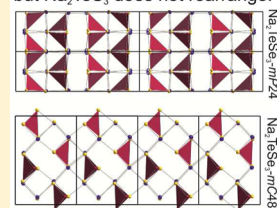
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**S** Supporting Information

**ABSTRACT:** Pure samples of Na<sub>2</sub>TeS<sub>3</sub> and Na<sub>2</sub>TeSe<sub>3</sub> were synthesized by the reactions of stoichiometric amounts of the elements Na, Te, and Q (Q = S, Se) in the ratio 2:1:3. Both compounds are highly air- and moisture-sensitive. The crystal structures were determined by single-crystal X-ray diffraction. Yellow Na<sub>2</sub>TeS<sub>3</sub> crystallizes in the space group *P2<sub>1</sub>/c*. Na<sub>2</sub>TeSe<sub>3</sub> exists in a low-temperature modification (Na<sub>2</sub>TeSe<sub>3</sub>-*mP24*, space group *P2<sub>1</sub>/c*) and a high-temperature modification (Na<sub>2</sub>TeSe<sub>3</sub>-*mC48*, space group *C2/c*); both modifications are red. Density functional theory calculations confirmed the coexistence of both modifications of Na<sub>2</sub>TeSe<sub>3</sub> because they are very close in energy ( $\Delta E = 0.18 \text{ kJ mol}^{-1}$ ). To the contrary, hypothetical Na<sub>2</sub>TeSe<sub>3</sub>-*mC48* is significantly less favored ( $\Delta E = 1.8 \text{ kJ mol}^{-1}$ ) than the primitive modification. Na<sub>2</sub>TeS<sub>3</sub> and Na<sub>2</sub>TeSe<sub>3</sub>-*mP24* are isotypic to Li<sub>2</sub>TeS<sub>3</sub>, whereas Na<sub>2</sub>TeSe<sub>3</sub>-*mC48* crystallizes in its own structure type, which was first described by Eisenmann and Zagler. The title compounds have two common structure motifs. Trigonal TeQ<sub>3</sub> pyramids form layers, and the Na atoms are surrounded by a distorted octahedral environment of chalcogen atoms. Raman spectra are dominated by the vibration modes of the TeQ<sub>3</sub> units. The activation energies of the total conductivity of the title compounds range between 0.68 eV (Na<sub>2</sub>TeS<sub>3</sub>) and 1.1 eV (Na<sub>2</sub>TeSe<sub>3</sub>). Direct principal band gaps of 1.20 and 1.72 eV were calculated for Na<sub>2</sub>TeSe<sub>3</sub> and Na<sub>2</sub>TeS<sub>3</sub>, respectively. The optical band gaps are in the range from 1.38 eV for Li<sub>2</sub>TeS<sub>3</sub> to 2.35 eV for Na<sub>2</sub>TeS<sub>3</sub>.

Na<sub>2</sub>TeSe<sub>3</sub> likes the change, but Na<sub>2</sub>TeS<sub>3</sub> does not rearrange.



## INTRODUCTION

Research on ternary chalcogenidotellurates started early in the 20th century. The first experiments on such compounds were limited on aqueous solutions. Gutbier and Flury investigated a solution of NaOH and tellurium sulfide, which was saturated with H<sub>2</sub>S. The observed precipitate was interpreted as the monohydrate Na<sub>2</sub>TeS<sub>3</sub>·H<sub>2</sub>O.<sup>1</sup> Alternatively, Te<sub>2</sub>O<sub>3</sub>(OH)NO<sub>3</sub> was used to synthesize numerous anhydrides and dihydrides of ternary sulfidotellurates(IV) with other mono- and bivalent cations in several investigations by Schäfer and co-workers. They precipitated K<sub>2</sub>TeS<sub>3</sub>, K<sub>3</sub>TeS<sub>4</sub>, Rb<sub>3</sub>TeS<sub>4</sub>, (NH<sub>4</sub>)<sub>2</sub>TeS<sub>3</sub>, (NH<sub>4</sub>)<sub>2</sub>TeS<sub>3</sub>·2H<sub>2</sub>O, and BaTeS<sub>3</sub>·2H<sub>2</sub>O by passing H<sub>2</sub>S in aqueous solutions of Te<sub>2</sub>O<sub>3</sub>(OH)NO<sub>3</sub> and KOH, respectively RbOH·2H<sub>2</sub>O, NH<sub>4</sub>OH, or Ba(OH)<sub>2</sub>.<sup>2</sup> No structure determinations were performed in the case of K<sub>3</sub>TeS<sub>4</sub> and Rb<sub>3</sub>TeS<sub>4</sub> to date, and therefore the nature of these two compounds is not yet clear. Further anhydrides of both thio- and selenotellurates(IV) were exclusively accessible by solid-state reactions. Jumas and co-workers obtained BaTeS<sub>3</sub> and Ba<sub>3</sub>Te<sub>2</sub>S<sub>7</sub> from mixtures of BaS, Te, and S, whereas Bensch and co-workers converted tellurium in a potassium polysulfide flux to K<sub>2</sub>TeS<sub>3</sub>.<sup>3</sup> Recently, Schleid and co-workers presented the heavier homologue Cs<sub>2</sub>TeS<sub>3</sub>.<sup>4</sup> Li<sub>2</sub>TeS<sub>3</sub> and Li<sub>2</sub>TeSe<sub>3</sub> were reported by Pfitzner and co-workers.<sup>5b,c</sup> Zagler and Eisenmann reported Na<sub>2</sub>TeSe<sub>3</sub>-*mC48*.<sup>5a</sup> These compounds were also obtained by solid-state reactions from the elements. In the literature, *mP24* modifications of Na<sub>2</sub>TeS<sub>3</sub> and Na<sub>2</sub>TeSe<sub>3</sub> were unknown to date.

The structures of these chalcogenidotellurates(IV) are mostly well determined. BaTeS<sub>3</sub>, BaTeS<sub>3</sub>·2H<sub>2</sub>O, and (NH<sub>4</sub>)<sub>2</sub>TeS<sub>3</sub> crystallize in orthorhombic space groups,<sup>2</sup> whereas the corresponding alkali-metal compounds crystallize in the monoclinic crystal system. Their structures have layers of trigonal-pyramidal TeQ<sub>3</sub> units (Q = S, Se) in common. The counteranions are located between these layers and are six-coordinated by the chalcogenide atoms. Li<sub>2</sub>TeQ<sub>3</sub> compounds (Q = S, Se) are isotypic and crystallize in their own structure type with a coordination number of 6 for Li.<sup>5b,c</sup> K<sub>2</sub>TeS<sub>3</sub> and Cs<sub>2</sub>TeS<sub>3</sub> show some similarities in comparison to these lithium compounds. However, their crystal structures differ from each other because of the different orientations of the TeS<sub>3</sub> units within the layers.<sup>3,4</sup>

Chalcogenidotellurates(IV) containing coinage metals besides alkali-metal cations are also known. Zhang and Kanatzidis reported new two-dimensional monoclinic structure types for isostructural RbCuTeS<sub>3</sub>, KAgTeS<sub>3</sub>, RbAgTeS<sub>3</sub>, and CsAgTeS<sub>3</sub>. Monovalent Ag and Cu ions are incorporated in the TeS<sub>3</sub> layers via tetrahedral coordination by S, and alkali-metal ions reside between the layers.<sup>6</sup>

A compound with the composition “Cu<sub>2</sub>TeS<sub>3</sub>” has not been reported to date. However, Pfitzner and co-workers were able to create layers of this composition, which were embedded in CuX (X = Cl, I) matrixes to form the compounds

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Table 1. Crystallographic Data of Na<sub>2</sub>TeS<sub>3</sub>, Na<sub>2</sub>TeSe<sub>3</sub>-mP24, and Na<sub>2</sub>TeSe<sub>3</sub>-mC48

	Na <sub>2</sub> TeS <sub>3</sub>	Na <sub>2</sub> TeSe <sub>3</sub> -mP24	Na <sub>2</sub> TeSe <sub>3</sub> -mC48
color	yellow	red	red
<i>M<sub>w</sub></i> /(g mol <sup>-1</sup> )	269.76	410.46	410.46
cryst syst	monoclinic	monoclinic	monoclinic
space group	<i>P</i> 2 <sub>1</sub> / <i>c</i> (No. 14)	<i>P</i> 2 <sub>1</sub> / <i>c</i> (No. 14)	<i>C</i> 2/ <i>c</i> (No. 15)
<i>a</i> /Å	5.776(1) <sup>a</sup>	5.990(1) <sup>a</sup>	21.6801(9), <sup>a</sup> 21.627(1) <sup>b</sup>
<i>b</i> /Å	12.202(3) <sup>a</sup>	12.658(3) <sup>a</sup>	5.9754(2), <sup>a</sup> 5.9730(2) <sup>b</sup>
<i>c</i> /Å	8.405(3) <sup>a</sup>	8.784(2) <sup>a</sup>	11.9736(7), <sup>a</sup> 11.9519(6) <sup>b</sup>
β/deg	92.02(2) <sup>a</sup>	93.51(3) <sup>a</sup>	121.166(6), <sup>a</sup> 121.228(7) <sup>b</sup>
<i>V</i> /Å <sup>3</sup>	592.0(3) <sup>a</sup>	664.8(2) <sup>a</sup>	1326.8(2), <sup>a</sup> 1320.3(1) <sup>b</sup>
<i>Z</i>	4	4	8
ρ <sub>calc</sub> /(g cm <sup>-3</sup> )	3.027	4.101	4.141, <sup>a</sup> 4.130 <sup>b</sup>
temperature <i>T</i> /K	293	293	293, <sup>a</sup> 123 <sup>b</sup>
diffractometer	STOE IPDS, graphite monochromator	STOE IPDS, graphite monochromator	Oxford Diff. Mova, graphite monochromator
λ(Mo <i>K</i> α)/Å	0.71073	0.71073	0.71073
μ(Mo <i>K</i> α)/mm <sup>-1</sup>	6.073	20.893	21.095
abs corr	numerical; crystal description with 7 faces; shape optimized with X-SHAPE <sup>12</sup>	numerical; crystal description with 19 faces; shape optimized with X-SHAPE <sup>12</sup>	analytical; Clark and Reid <sup>13</sup> crystal description optimized with SCALE ABSPACK <sup>13</sup>
2θ range/deg	5.88 ≤ 2θ ≤ 51	5.66 ≤ 2θ ≤ 52.16	6.82 ≤ 2θ ≤ 51
<i>hkl</i> ranges	-7 ≤ <i>h</i> ≤ 7 -16 ≤ <i>k</i> ≤ 16 -10 ≤ <i>l</i> ≤ 9	-7 ≤ <i>h</i> ≤ 7 -15 ≤ <i>k</i> ≤ 15 -10 ≤ <i>l</i> ≤ 10	-31 ≤ <i>h</i> ≤ 31 -5 ≤ <i>k</i> ≤ 8 -17 ≤ <i>l</i> ≤ 17
no. of param	55	55	55
no. of reflns, <i>R</i> <sub>int</sub>	5996, 0.0325	4729, 0.0384	6942, 0.0387
no. of indep reflns	1381	1308	2147
final <i>R</i> <sub>1</sub> , <i>wR</i> <sub>2</sub> [ <i>I</i> > 2σ( <i>I</i> )]	0.0175, 0.0463	0.0239, 0.0621	0.0352, 0.0745
final <i>R</i> <sub>1</sub> , <i>wR</i> <sub>2</sub> (all reflns)	0.0208, 0.0451	0.0272, 0.0608	0.0326, 0.0730
GOF	1.015	0.981	1.080
largest diff peak Δρ <sub>max</sub> /hole Δρ <sub>min</sub> /(e Å <sup>-3</sup> )	0.557, -0.432	1.03, -0.701	1.363, -2.726

<sup>a</sup>Parameters refined from powder X-ray diffraction data at room temperature. <sup>b</sup>Parameters refined from single-crystal diffraction data at 123 K.

CuClCu<sub>2</sub>TeS<sub>3</sub> and (CuI)<sub>3</sub>Cu<sub>2</sub>TeS<sub>3</sub>. Therein TeS<sub>3</sub><sup>2-</sup> units are also arranged in layers, and Cu occupies both trigonal-planar and tetrahedral sites.<sup>7</sup> Kanatzidis and co-workers observed the corresponding Ag-containing layers Ag<sub>2</sub>TeS<sub>3</sub> in the compounds Ag<sub>2</sub>TeS<sub>3</sub>·Rb<sub>2</sub>S<sub>6</sub> and Ag<sub>2</sub>TeS<sub>3</sub>·Cs<sub>2</sub>S<sub>6</sub>, in which they are separated by polysulfide chains S<sub>6</sub><sup>2-</sup>.<sup>8</sup> Furthermore, pure, monoclinic Ag<sub>2</sub>TeS<sub>3</sub> (space group *Cc*) was presented by Pertlik.<sup>9</sup> The tetrahedral coordination of Ag remains, even when 75% of Ag is replaced by Na. The resulting compound has the composition Na<sub>1.5</sub>Ag<sub>0.5</sub>TeS<sub>3</sub> with a 1:1 mixing of Ag and Na on one of two Ag sites. A complete substitution of Ag by Na results in the title phase Na<sub>2</sub>TeS<sub>3</sub>, which was reported by Preitschaft.<sup>5b</sup> He found that Na<sub>2</sub>TeS<sub>3</sub> and Na<sub>2</sub>TeSe<sub>3</sub> are isotypic with Li<sub>2</sub>TeQ<sub>3</sub>, which crystallizes in the space group *P*2<sub>1</sub>/*c*.<sup>5b,c</sup> However, Zagler and Eisenmann proposed a different structure type for Na<sub>2</sub>TeSe<sub>3</sub> with the space group *C*2/*c* in an earlier communication.<sup>5a</sup>

Herein, we present the first sodium thiotellurate(IV) and a new phase of Na<sub>2</sub>TeSe<sub>3</sub>. The structural results are interpreted by taking Raman spectroscopy, thermal analysis, and density functional theory (DFT) calculations into account. Also, the question of why two different modifications are observed for the selenide but only one for the sulfide is discussed.

## EXPERIMENTAL SECTION

**General Synthetic Procedure.** All samples were synthesized and manipulated in a glovebox under an argon atmosphere. Na<sub>2</sub>TeS<sub>3</sub> and Na<sub>2</sub>TeSe<sub>3</sub> were synthesized from the elements Na (99%, Merck), Te (99.999%, Chempur), S (99.999%, Chempur), and Se (99.99+%,

Chempur), respectively, in the stoichiometric ratio 2:1:3. The starting materials were filled in graphite crucibles, which were sealed in evacuated silica ampules. The samples of Na<sub>2</sub>TeS<sub>3</sub> and Na<sub>2</sub>TeSe<sub>3</sub>-mC48 were heated up to 1170 K (1 K min<sup>-1</sup>) in order to obtain well-crystalline material. After 3 h, they were slowly cooled to room temperature (0.1 K min<sup>-1</sup>). Na<sub>2</sub>TeSe<sub>3</sub>-mP24 was obtained by annealing the elements at 670 K for 10 days. The lithium compounds were synthesized as described in ref 5c.

**Powder X-ray Diffraction.** X-ray diffraction on powdered samples was performed with a Huber diffractometer (Cu *K*α<sub>1</sub> radiation; λ = 1.540598 Å; Ge(111) monochromator; imaging-plate camera system G670) to prove the purity of the samples. They were fixed between two mylar films (Mylar Polyester, Chempur). A thin film of mineral oil prevented decomposition of the compounds by air. Data were collected in a 2θ range of 10–80° with a step size of 0.005°. The STOE WINXPOW program package was used for indexing and cell refinements.<sup>10</sup> The cell parameters were finally refined with JANA2006.<sup>11</sup>

**Crystal Structure Determination.** Data for the crystal structure determination of Na<sub>2</sub>TeS<sub>3</sub> and Na<sub>2</sub>TeSe<sub>3</sub>-mP24 were obtained by single-crystal X-ray diffraction (Stoe IPDS I; graphite monochromator; Mo *K*α radiation; λ = 0.71073 Å). The crystals were fixed on top of glass fibers and measured in sealed glass capillaries at room temperature. Absorption was corrected numerically by X-RED after optimization of the crystal-shape description with X-SHAPE.<sup>12</sup> A single crystal of Na<sub>2</sub>TeSe<sub>3</sub>-mC48 was measured additionally at low temperatures on an Oxford CCD diffractometer equipped with a MOVA microsource (Mo *K*α radiation; λ = 0.71073 Å). A Cryojet unit maintained a sample temperature of 123 K. The crystal was fixed on a microloop in mineral oil. The crystal-shape description was optimized with SCALE3 ABSPACK, and an analytical absorption correction according to Clark and Reid was applied.<sup>13</sup> The structure models of

the title compounds were solved with *SHELXS97* by direct methods and refined on the basis of full-matrix least squares with *SHELXL97*.<sup>14</sup>

**Differential Thermal Analysis (DTA).** The measurements were performed with a Setaram DTA-TG 92-16.18 with  $\text{Al}_2\text{O}_3$  as the standard. Small amounts of the title compounds were sealed in evacuated quartz tubes (1.5 mm in diameter). DTA data were recorded up to 1073 K with heating and cooling rates of 10 K  $\text{min}^{-1}$ .

**Raman Spectroscopy.** Raman spectra were recorded with a Varian FTS 7000e spectrometer with a Nd:YAG laser (excitation wavelength  $\lambda = 1064 \text{ \AA}$ ), a Varian FT-Raman module, and a liquid-nitrogen-cooled germanium detector. Data were collected in back-scattering mode. The powder samples were sealed in Duran glass capillaries (external diameter 1.5 mm).

**Theoretical Calculations.** All quantum-chemical calculations were performed within the DFT-generalized gradient approximation (GGA) functional Perdew–Burke–Ernzerhof.<sup>15</sup> Full geometry optimizations were performed with the projector-augmented-wave<sup>16</sup> approach and the conjugant gradient algorithm, as implemented in the code of the Vienna ab initio simulation package (*VASP 4.6*);<sup>16</sup> see also ref 17. Convergence is considered at differences in the total energy of less than  $10^{-5} \text{ eV}$  and maximum Hellmann–Feynman forces of  $10^{-4} \text{ eV \AA}^{-1}$ . Density of states (DOS) and band-structure calculations were executed with the scalar relativistic full-potential local orbital scheme FPLO14, which optimizes valence functions at each self-consistent-field step.<sup>18a,b</sup> The calculations were converged to  $4 \times 4 \times 4$   $k$ -point meshes.

**UV/Vis Spectroscopy.** For the optical-band-gap determination, the samples were diluted with  $\text{BaSO}_4$  (pure Ph. Eur. AppliChem) and measured with a Bruins Omega 20 spectrometer in remission mode with a resolution of 1 nm. The wavelength of the emitted light ranged between 380 and 1100 nm. The instrument was operated by the software *OMEGA Analyzer*.<sup>19a</sup> 100% remission was defined for the pure standard  $\text{BaSO}_4$ . The remission was converted to absorption using the Kubelka–Munk function.<sup>19b,c</sup> The optical band gap was assumed as the intersection of linear extrapolations of the baseline and absorption edge.

**Impedance Spectroscopy.** Complex impedance spectra were recorded with an IM6 analyzer (Zahner Elektrik, Meßtechnik, Germany) under an argon atmosphere. Therefore, finely ground powders were pressed to pellets with at least 90% of the crystallographic density. The pellets were contacted to Au electrodes and transferred to a homemade measuring cell. Spectra were recorded in the range from 1 Hz to 1 MHz. The *Thales Flink* software was used for operation and data evaluation.<sup>20</sup> A detailed description of the setup is given in ref 21.

## RESULTS AND DISCUSSION

**Crystal Structure of  $\text{Na}_2\text{TeS}_3$ .**  $\text{Na}_2\text{TeS}_3$  crystallizes in the space group  $P2_1/c$  with the lattice parameters  $a = 5.776(1) \text{ \AA}$ ,  $b = 12.202(3) \text{ \AA}$ ,  $c = 8.405(3) \text{ \AA}$ ,  $\beta = 92.02(2)^\circ$ , and  $V = 592.0(3) \text{ \AA}^3$  ( $Z = 4$ ). Crystallographic details are listed in Table 1.  $\text{Na}_2\text{TeS}_3$  is isotypic with  $\text{Li}_2\text{TeS}_3$ . For a better comparison, we chose the same crystallographic settings in Table 2 as those

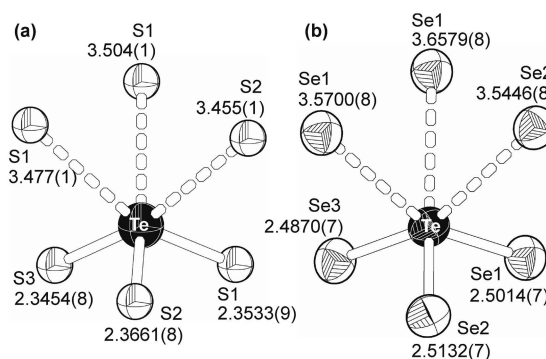
**Table 2. Atomic Coordinates and Equivalent Isotropic Displacement Parameters  $U_{\text{eq}}/\text{\AA}^2$  for  $\text{Na}_2\text{TeS}_3$  at Room Temperature**

atom <sup>a</sup>	<i>x</i>	<i>y</i>	<i>z</i>	$U_{\text{eq}}$
Na1	0.7402(2)	0.76002(9)	0.1240(2)	0.0294(2)
Na2	0.2458(2)	0.91664(8)	0.3641(2)	0.0273(2)
Te	0.70928(2)	0.08813(2)	0.18461(2)	0.01804(7)
S1	0.3051(1)	0.09304(5)	0.13899(9)	0.0219(1)
S2	0.7612(1)	0.23360(5)	0.36856(8)	0.0216(1)
S3	0.7482(1)	0.93561(5)	0.35327(8)	0.0227(1)

<sup>a</sup>All atoms are located on the Wyckoff site 4e.

in ref 5a. All atoms reside on the Wyckoff position 4e. The sulfide ion sites S1, S2, and S3 constitute a distorted cubic-close-packed anionic framework with a total of 12 octahedral voids per unit cell. Therein,  $\text{Te}^{\text{IV}}$  cations occupy one-third of crystallographic equivalent octahedral sites, with a strong 3 + 3 splitting of the distances  $d(\text{Te}–\text{S})$ .

Te atoms have three S atoms in a distance of about 2.4  $\text{\AA}$  and three additional S atoms in larger distances of about 3.5  $\text{\AA}$ . This is due to the predominantly covalent bonding character of the Te–S bonds. The 3 + 3 coordination of Te is shown in Figure 1a. The three short distances  $d(\text{Te}–\text{S})$  in these isolated trigonal



**Figure 1.** 3 + 3 coordination of Te by the chalcogenide anions in (a)  $\text{Na}_2\text{TeS}_3$  and (b)  $\text{Na}_2\text{TeSe}_3$ -mP24 and the interatomic distances  $d(\text{Te}–\text{S})$  and  $d(\text{Te}–\text{Se})$  given in angstroms.

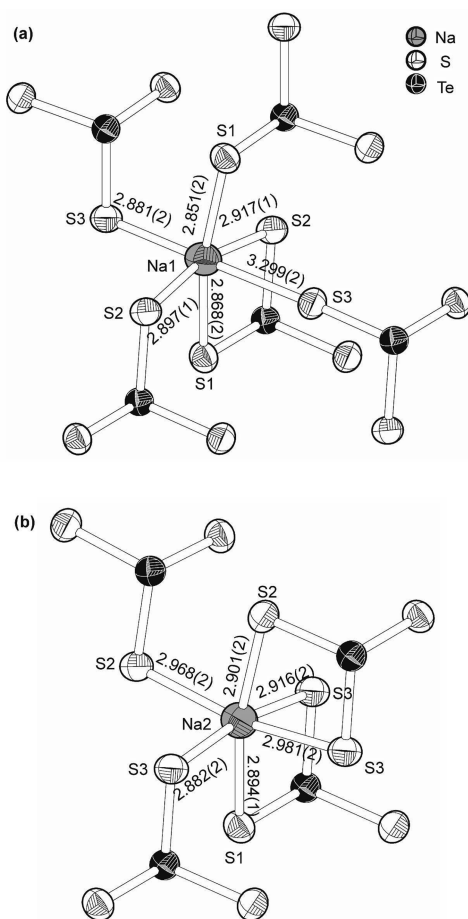
$\text{TeS}_3$  pyramids range from 2.345 to 2.366  $\text{\AA}$ , as was already observed for similar compounds. For example, an average value of  $\bar{d}(\text{Te}–\text{S}) = 2.367 \text{ \AA}$  was reported for  $\text{Li}_2\text{TeS}_3$  and  $\bar{d}(\text{Te}–\text{S}) = 2.350 \text{ \AA}$  for  $\text{K}_2\text{TeS}_3$ .<sup>3,5c</sup> The smallest distances  $d(\text{Te}–\text{S})$  in this series of alkali-metal compounds were recently found in  $\text{Cs}_2\text{TeS}_3$  (2.312–2.349  $\text{\AA}$ ).<sup>3</sup>

The size of the alkali metal M has only a minor influence on the distance  $d(\text{Te}–\text{S})$ . The values slightly decrease with increasing radius and increasing electropositivity of M. An opposite trend is found for the angles S–Te–S in this series of compounds. They become larger for the heavier alkali-metal thiotellurates. Thus, they range between  $100.9^\circ$  and  $101.1^\circ$  in  $\text{Na}_2\text{TeS}_3$ . They are larger than those in  $\text{Li}_2\text{TeS}_3$  ( $99.2$ – $100.5^\circ$ ) and smaller than those in  $\text{K}_2\text{TeS}_3$  ( $102.1$ – $103.1^\circ$ ) and  $\text{Cs}_2\text{TeS}_3$  ( $102.7$ – $106.9^\circ$ ).<sup>3–5</sup>

The Na cations occupy two-thirds of the distorted octahedral voids in the sulfide substructure. Figure 2 shows such  $\text{NaS}_6$  motifs for the two different sites Na1 and Na2. It is remarkable that the distance  $d(\text{Na1}–\text{S3}) = 3.299 \text{ \AA}$  deviates significantly from all other distances  $d(\text{Na}–\text{S})$ , which are considerably smaller than 3  $\text{\AA}$ . This specific distance is about 15% longer than the sum of the ionic radii (2.86  $\text{\AA}$ ).<sup>22</sup> Therefore, this surrounding is better described as a 5 + 1 coordination. In contrast, the distances  $d(\text{Na2}–\text{S})$  are more or less in the range of the sum of the ionic radii. All six values for  $d(\text{Na2}–\text{S})$  range from 2.882 to 2.981(2)  $\text{\AA}$ . This effect is even more pronounced in the case of the isotypic compound  $\text{Li}_2\text{TeS}_3$ .<sup>5c</sup>

This structural peculiarity is also observed in chalcogenidotellurates, which are not isotypic to  $\text{Na}_2\text{TeS}_3$ . In  $\text{K}_2\text{TeS}_3$ , for example, one longer distance  $d(\text{K}–\text{S}) = 3.536 \text{ \AA}$  and an average value of  $\bar{d}(\text{K}–\text{S}) = 3.22 \text{ \AA}$  for the other distances are observed.<sup>3</sup> The angles S–Na–S are almost unaffected by this one elongated interatomic distance. They range from  $78.0$  to  $177.6^\circ$  for S–Na1–S and from  $76.4$  to  $170.2^\circ$  for S–Na2–S.





**Figure 2.** Different coordination of the sites (a) Na1 and (b) Na2 by the trigonal  $\text{TeSe}_3$  pyramids, which act as mono- or bidentate ligands.

Table S2 summarizes the corresponding data. Apart from these distortions, both Na sites show also a difference with respect to the coordination mode of the  $\text{TeSe}_3$  units; see Figure 2.  $\text{Na1Se}_6$  octahedra are connected to four monodentate  $\text{TeSe}_3$  ligands and one bidentate  $\text{TeSe}_3$  ligand.  $\text{Na2Se}_6$  octahedra are connected to two monodentate and two bidentate  $\text{TeSe}_3$  ligands. Multidentate  $\text{TeSe}_3$  ligands are also known from other structure types like, e.g.,  $\text{Cs}_2\text{TeSe}_3$ .<sup>4</sup>

**Crystal Structure of  $\text{Na}_2\text{TeSe}_3$ -mP24.** The red compound  $\text{Na}_2\text{TeSe}_3$ -mP24 is isotypic to  $\text{Na}_2\text{TeSe}_3$  and has the lattice parameters  $a = 5.990(1)$  Å,  $b = 12.658(3)$  Å,  $c = 8.784(2)$  Å,  $\beta = 93.51(3)^\circ$ , and  $V = 664.8(2)$  Å<sup>3</sup> ( $Z = 4$ ), which were refined from powder X-ray diffraction data (Table 3).

**Table 3. Atomic Coordinates and Equivalent Isotropic Displacement Parameters  $U_{\text{eq}}/\text{\AA}^2$  for  $\text{Na}_2\text{TeSe}_3$ -mP24 at Room Temperature**

atom <sup>a</sup>	<i>x</i>	<i>y</i>	<i>z</i>	$U_{\text{eq}}$
Na1	0.7389(3)	0.7616(2)	0.1296(2)	0.0382(4)
Na2	0.2461(3)	0.9167(1)	0.3627(2)	0.0362(4)
Te	0.70263(4)	0.08924(2)	0.18948(3)	0.0260(1)
Se1	0.28807(7)	0.09232(2)	0.13406(5)	0.0302(1)
Se2	0.76189(7)	0.23828(3)	0.37661(5)	0.0293(1)
Se3	0.74859(7)	0.93291(4)	0.36001(5)	0.0307(1)

<sup>a</sup>All atoms are positioned on the Wyckoff site 4e.

The distances  $d(\text{Te}-\text{Se})$  suggest the motif of isolated  $\text{TeSe}_3$  units. Three short distances are between 2.4869 and 2.5131 Å, whereas the distances to the second-nearest neighbors range from 3.545 to 3.658 Å. A comparison of these direct contacts and the mean distances to next-nearest neighbors  $\bar{d}_{\text{sec}}(\text{Te}-\text{Se})$  in the compounds  $\text{Li}_2\text{TeSe}_3$  (2.513 Å) or  $\text{K}_2\text{TeSe}_3$  (2.491 Å) and the mean distances  $\bar{d}_{\text{sec}}(\text{Te}-\text{Se})$  in  $\text{M}_2\text{TeSe}_3$  show a reduction of  $d$  with increasing size and electropositivity of the cation.<sup>3,5</sup>

The distortion of the octahedron  $\text{Na2Se}_6$  is less pronounced than that of the octahedron  $\text{Na1Se}_6$ . All values of  $d(\text{Na2}-\text{Se})$  range between 2.986 and 3.090 Å. For  $d(\text{Na1}-\text{Se})$ , five values are between 2.960 and 3.011 Å. The shorter bond lengths are comparable to those in  $\text{Na}_3\text{SbSe}_3$  (2.992–3.218 Å).<sup>23</sup> The largest value in  $\text{Na}_2\text{TeSe}_3$ -mP24 is observed for  $d(\text{Na1}-\text{Se3}) = 3.419$  Å. Compounds with primary coordination spheres with  $d(\text{Na}-\text{Se}) > 3.4$  Å are quite rare. Ohtani and co-workers reported similar values (3.437 and 3.500 Å) for the compounds  $\text{NaTM}_5\text{Se}_8$  ( $\text{TM} = \text{Ti}, \text{Cr}$ ).<sup>24</sup> Details of the three-dimensional linkage of the building blocks are the same as those for  $\text{Na}_2\text{TeSe}_3$ -mP24 vide supra.

**Crystal Structure of  $\text{Na}_2\text{TeSe}_3$ -mC48.** The crystal structure of  $\text{Na}_2\text{TeSe}_3$ -mC48 was first determined by Zagler and Eisenmann.<sup>5a</sup> However, we herein present the results of a new structure determination at low temperature, which confirms their results. In addition, structural correlations between both phases of  $\text{Na}_2\text{TeSe}_3$  and their thermal transformation are discussed.  $\text{Na}_2\text{TeSe}_3$ -mC48 crystallizes in the space group  $C2/c$  with  $a = 21.627(1)$  Å,  $b = 5.9730(2)$  Å,  $c = 11.9519(6)$  Å,  $\beta = 121.228(7)^\circ$ , and  $V = 1320.3(1)$  Å<sup>3</sup> ( $Z = 8$ ) (data collected at 123 K). The cell parameters of  $\text{Na}_2\text{TeSe}_3$ -mC48 were refined from powder X-ray diffraction data with JANA2006 to  $a = 21.6801(9)$  Å,  $b = 5.9730(2)$  Å,  $c = 11.9519(6)$  Å,  $\beta = 121.228(7)^\circ$ , and  $V = 1326.8(2)$  Å<sup>3</sup> at room temperature. They are in good agreement with the data of Zagler and Eisenmann.<sup>5a</sup> For a better comparison, we chose their setup of atomic parameters (see Table 4). Both

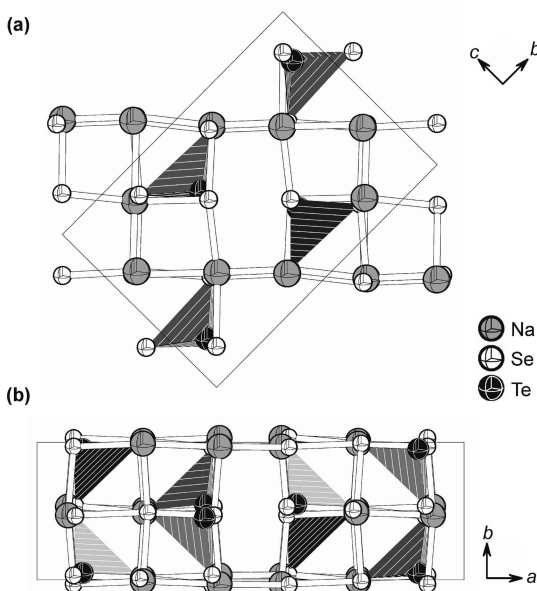
**Table 4. Atomic Coordinates and Equivalent Isotropic Displacement Parameters  $U_{\text{eq}}/\text{\AA}^2$  for  $\text{Na}_2\text{TeSe}_3$ -mC48 at Room Temperature**

atom <sup>a</sup>	<i>x</i>	<i>y</i>	<i>z</i>	$U_{\text{eq}}$
Na1	0.75184(9)	0.0108(3)	0.1246(2)	0.0164(4)
Na2	0.9274(1)	0.4686(2)	0.8076(2)	0.0178(4)
Te	0.89304(2)	0.04964(4)	0.49288(2)	0.01134(8)
Se1	0.91021(2)	0.46339(6)	0.54187(4)	0.0130(1)
Se2	0.91422(2)	0.03137(6)	0.30581(4)	0.0134(1)
Se3	0.75919(2)	0.00938(6)	0.38250(4)	0.0128(1)

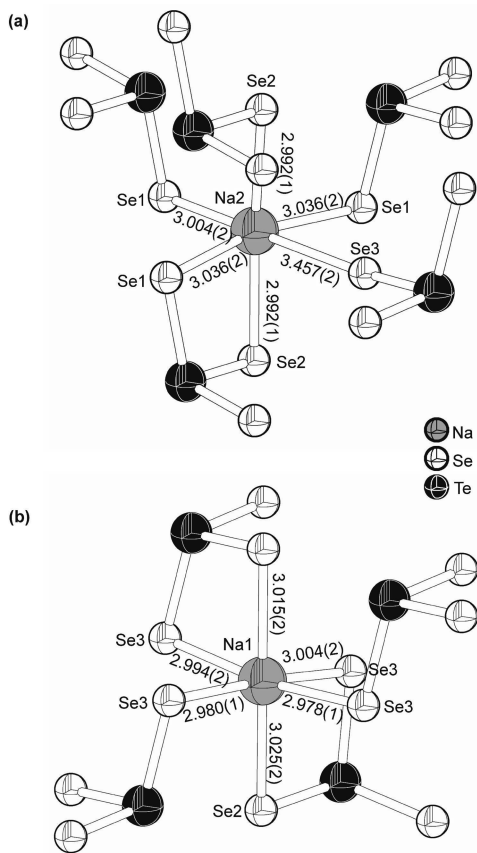
<sup>a</sup>All atoms are positioned on the Wyckoff site 8f.

modifications of  $\text{Na}_2\text{TeSe}_3$  have the already mentioned cubic-closed-packed framework of selenide anions in common. The difference between the structures is due to the way how the octahedral voids are occupied by sodium and tellurium. The two different ordering schemes of cations are shown in Figure 3. Because the unit cell volume of  $\text{Na}_2\text{TeSe}_3$ -mC48 is twice that of  $\text{Na}_2\text{TeSe}_3$ -mP24, the cell has 24 occupied octahedral sites.

The crystal structure of  $\text{Na}_2\text{TeSe}_3$ -mC48 consists of the same basic building units as the second modification  $\text{Na}_2\text{TeSe}_3$ -mP24, trigonal-pyramidal  $\text{TeSe}_3$  units and two octahedrally coordinated Na sites (Figure 4). Again, one of these  $\text{NaSe}_6$  octahedra shows one pronounced long distance  $d(\text{Na2}-\text{Se}) =$



**Figure 3.** Ordering schemes of Te and Na in the cubic-closest-packed framework of Se anions in the case of (a)  $\text{Na}_2\text{TeSe}_3\text{-mP24}$  and (b)  $\text{Na}_2\text{TeSe}_3\text{-mC48}$ . The  $\text{TeSe}_3$  pyramids are illustrated in dark gray.



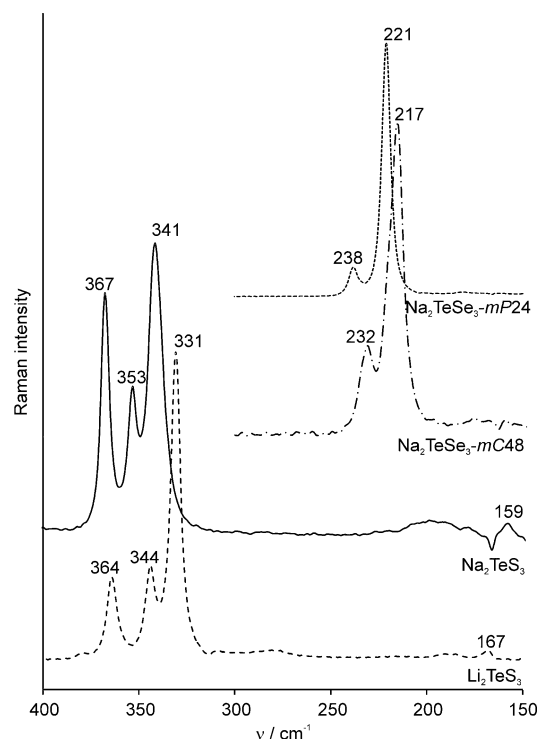
**Figure 4.** Trigonal  $\text{TeSe}_3$  pyramids act as mono- or bidentate ligands in  $\text{Na}_2\text{TeSe}_3\text{-mC48}$  similar to that in  $\text{Na}_2\text{TeSe}_3\text{-mP24}$ .

3.457(2) Å. It is connected to four monodentate  $\text{TeSe}_3$  ligands and one bidentate  $\text{TeSe}_3$  ligand, whereas the less distorted octahedron is connected to two monodentate and two bidentate  $\text{TeSe}_3$  ligands. The interatomic distances are

generally almost identical for both phases of  $\text{Na}_2\text{TeSe}_3$  (compare Tables S4 and S6).

**Raman Spectroscopy.** To date, vibration modes of the  $\text{TeSe}_3$  and  $\text{TeSe}_2$  units were observed only for a few compounds. In the case of thiotellurates(IV), for example, the Raman shifts  $\nu(\text{Te-S})$  for  $\text{Cs}_2\text{TeS}_3$  (392, 370, and  $192\text{ cm}^{-1}$ ),<sup>4</sup>  $\text{Tl}_2\text{TeS}_3$  (338, 309, 180, and  $154\text{ cm}^{-1}$ ),<sup>25</sup>  $\text{CuClCu}_2\text{TeS}_3$  (359, 336, 274, 247, and  $155\text{ cm}^{-1}$ ), and  $(\text{CuI})_3\text{Cu}_2\text{TeS}_3$  (371, 360, 328, and  $108\text{ cm}^{-1}$ ) were reported.<sup>7</sup> They resemble each other in the number of detected signals and their intensity ratio because of their similar local symmetry. An ideal  $C_{3v}$  symmetry of the  $\text{TeSe}_3$  units was observed in  $\text{CuClCu}_2\text{TeS}_3$ .<sup>7</sup> A crystallographic symmetry reduction resulting in a structural distortion of the  $\text{TeSe}_3$  pyramids leads to the point groups  $C_s$  (as reported for  $\text{Tl}_2\text{TeS}_3$ ) and  $C_1$  like in  $(\text{CuI})_3\text{Cu}_2\text{TeS}_3$ . The latter holds also for the title compounds.

For  $\text{Na}_2\text{TeSe}_3$ , we observed signals at 367, 353, 341, and  $159\text{ cm}^{-1}$  (see Figure 5). This corresponds to the values reported

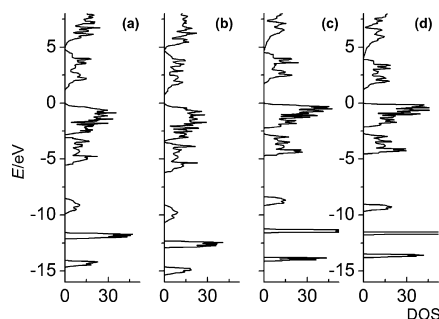


**Figure 5.** Raman spectra of  $\text{Li}_2\text{TeS}_3$  (broken line),  $\text{Na}_2\text{TeS}_3$  (solid line),  $\text{Na}_2\text{TeSe}_3\text{-mP24}$  (dotted line), and  $\text{Na}_2\text{TeSe}_3\text{-mC48}$  (broken-dotted line).

for  $(\text{CuI})_3\text{Cu}_2\text{TeS}_3$ . The different Raman shifts can be explained by the varying values of  $d_{\text{sec}}(\text{Te-S})$  in these compounds. The longer  $d_{\text{sec}}(\text{Te-S})$ , the stronger are the three short bonds and the higher are the Raman resonances. For this reason, the Raman spectra of the isotopic compound  $\text{Li}_2\text{TeS}_3$  (broken line) differs slightly from that of  $\text{Na}_2\text{TeS}_3$ . The signals are red-shifted to values of 364, 344, 331, and  $167\text{ cm}^{-1}$ . The mean distances to the next-nearest neighbors  $\bar{d}_{\text{sec}}(\text{Te-S})$  in  $\text{Na}_2\text{TeS}_3$  are significantly longer than those in  $\text{Li}_2\text{TeS}_3$ , whereas in both compounds, the three short distances  $d(\text{Te-S})$  are more or less identical. The average of  $\bar{d}_{\text{sec}}(\text{Te-S})$  in  $\text{Li}_2\text{TeS}_3$  is  $3.200\text{ Å}$ , and  $\bar{d} = 3.479\text{ Å}$  in  $\text{Na}_2\text{TeS}_3$ . This is attributed to an increasing ionicity of the compound due to substitution of Li by Na. In the case of  $\text{K}_2\text{TeS}_3$ , a mean distance of  $\bar{d}_{\text{sec}}(\text{Te-S}) =$

3.928 Å is observed. However, no Raman shifts were reported for this compound, but IR studies were published by Zhang and Kanatzidis. They reported the strongest far-IR shifts at 371, 347, 210, and 190 cm<sup>-1</sup>.<sup>6</sup> Although the selection rules differ for IR and Raman spectroscopy, the shifts in these spectra may be taken for comparison. For the heaviest homologue Cs<sub>2</sub>TeS<sub>3</sub>, Schleid and co-workers observed the Te–S Raman modes at 392, 370, 192, and 176 cm<sup>-1</sup>.<sup>4</sup> Considering  $\bar{d}_{\text{sec}}$ (Te–S) is longer than 4 Å in Cs<sub>2</sub>TeS<sub>3</sub>, these shifts confirm the described trend. Kysliak and Beck observed Raman modes between 260 and 218 cm<sup>-1</sup> for [Zn(NH<sub>3</sub>)<sub>4</sub>]<sub>2</sub>TeSe<sub>3</sub>.<sup>26</sup> The spectra of the *mp24* and *mC48* modifications of Na<sub>2</sub>TeSe<sub>3</sub> show signals from 238 to 217 cm<sup>-1</sup>; see Figure 5. The surroundings of Te in these modifications are almost equal, which explains the similarity of the spectra.

**Theoretical Calculations.** At first, DFT calculations focused on the polymorphism of Na<sub>2</sub>TeSe<sub>3</sub> and the attempt to understand why no phase transition is observed for Na<sub>2</sub>TeSe<sub>3</sub>. Total energy calculations were performed on completely optimized structures to elucidate the relative stabilities of both polymorphs (Figure 6). In addition, a hypothetical,



**Figure 6.** Calculated total DOS of the isotopic compounds (a) Li<sub>2</sub>TeS<sub>3</sub>, (b) Li<sub>2</sub>TeSe<sub>3</sub>, (c) Na<sub>2</sub>TeS<sub>3</sub>, and (d) Na<sub>2</sub>TeSe<sub>3</sub>-*mp24*.

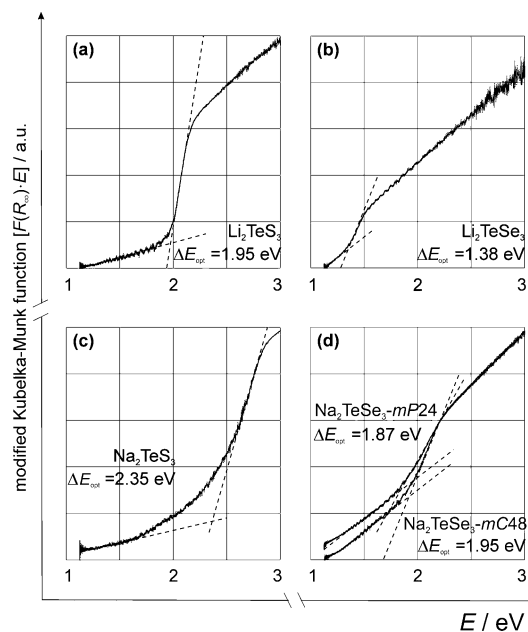
isostructural modification Na<sub>2</sub>TeS<sub>3</sub>-*mC48* was estimated. Predicted cell volumes are overestimated compared to the experiments, as is known for GGA methods, but the deviations are similar for both modifications. A difference in energy of  $\Delta E = 1.8$  kJ mol<sup>-1</sup> is calculated for hypothetical Na<sub>2</sub>TeS<sub>3</sub>-*mC48* compared to the modification *mp24*. This difference is smaller by 1 order of magnitude for the *mp24* and *mC48* modifications of Na<sub>2</sub>TeSe<sub>3</sub> ( $\Delta E = 0.18$  kJ mol<sup>-1</sup>). Such differences were identified as decisive whether structures are formed under equilibrium conditions in recent systematic investigations on the phase stabilities.<sup>17</sup> From the present calculations, Na<sub>2</sub>TeS<sub>3</sub>-*mC48* is predicted as less stable on the potential energy surface (0 K). However, it might be accessible by nonequilibrium methods. Its predicted crystal structure shows the same features as those of Na<sub>2</sub>TeSe<sub>3</sub>-*mC48*. A detailed analysis is, however, beyond the scope of the present paper.

The calculated electronic band structures of the title compounds are compared to Li<sub>2</sub>TeQ<sub>3</sub> (Q = S, Se). One can find a systematic trend in the change of the optical band gaps with the composition of the *mp24* modifications (Table 5, Figure 7, and Figure S2). The calculated band gap (DFT-GGA) decreases in accordance with the experimentally found colors when S is replaced by Se from 1.71 eV (Na<sub>2</sub>TeS<sub>3</sub>) to 1.18 eV (Na<sub>2</sub>TeSe<sub>3</sub>). When Na is replaced by Li, even smaller band gaps of 1.20 and 0.73 eV are calculated for Li<sub>2</sub>TeS<sub>3</sub> and Li<sub>2</sub>TeSe<sub>3</sub>, respectively. Interestingly, the predicted gap slightly increases from the *mp24* and *mC48* modifications (1.20 eV),

**Table 5.** Calculated Electronic Band Gaps  $\Delta E^{\text{calc}}$  of M<sub>2</sub>TeQ<sub>3</sub> from DFT-GGA in Comparison to the Experimentally Determined Optical Band Gap  $\Delta E^{\text{exp}}_{\text{opt}}$

	$\Delta E^{\text{calc}}/\text{eV}$	$\Delta E^{\text{exp}}_{\text{opt}}/\text{eV}$
Li <sub>2</sub> TeS <sub>3</sub>	1.197	1.95
Li <sub>2</sub> TeSe <sub>3</sub>	0.727	1.38
Na <sub>2</sub> TeS <sub>3</sub> - <i>mp24</i>	1.714	2.35
Na <sub>2</sub> TeS <sub>3</sub> - <i>mC48</i> <sup>a</sup>	1.468	
Na <sub>2</sub> TeSe <sub>3</sub> - <i>mp24</i>	1.179	1.87
Na <sub>2</sub> TeSe <sub>3</sub> - <i>mC48</i>	1.203	1.95

<sup>a</sup>Hypothetical modification.



**Figure 7.** Optical absorption spectra of M<sub>2</sub>TeQ<sub>3</sub> (M = Li, Na; Q = S, Se). The spectra of both modifications of Na<sub>2</sub>TeSe<sub>3</sub> are displayed in part d and are almost identical.

whereas it is significantly lowered for the hypothetical Na<sub>2</sub>TeS<sub>3</sub>-*mC48* (1.47 eV). The band-structure plots in Figure S2 of the title compounds indicate direct band gaps (see Figure S2). The underestimation of the absolute values is a well-known effect of DFT functionals.

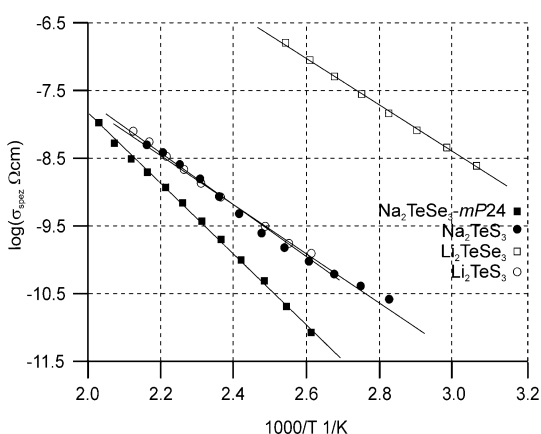
**UV/Vis Spectroscopy.** We carried out UV/vis spectroscopic measurements on Li<sub>2</sub>TeQ<sub>3</sub>, Na<sub>2</sub>TeS<sub>3</sub>, and both modifications of Na<sub>2</sub>TeSe<sub>3</sub> in order to verify the results of DFT calculations; see Figure 7. The detected optical band gap range between 1.38 and 2.35 eV. The smallest optical band gap was measured for dark-gray Li<sub>2</sub>TeSe<sub>3</sub> and the largest band gap for yellow Na<sub>2</sub>TeS<sub>3</sub>. The other samples show band gaps in the medium range of Li<sub>2</sub>TeS<sub>3</sub> (1.95 eV), Na<sub>2</sub>TeSe<sub>3</sub>-*mC48* (1.95 eV), and Na<sub>2</sub>TeSe<sub>3</sub>-*mp24* (1.87 eV).

The optical band gaps of tellurates(IV) under discussion increase with increasing ionicity of the alkali metals and chalcogens. The band gap of Li<sub>2</sub>TeS<sub>3</sub> is 0.40 eV smaller than that of Na<sub>2</sub>TeS<sub>3</sub>. Likewise, the alkali cation influences the band gap of the selenide compounds: The band gap of Li<sub>2</sub>TeSe<sub>3</sub> is 0.49 eV smaller than the one of Na<sub>2</sub>TeSe<sub>3</sub>. The influence of the chalcogenide on the optical properties is slightly stronger. Thus, the band gap of Li<sub>2</sub>TeSe<sub>3</sub> is 0.57 eV smaller than that of Li<sub>2</sub>TeS<sub>3</sub>. A similar relationship is observed for the sodium compounds. The optical band gap of Na<sub>2</sub>TeSe<sub>3</sub> is 0.48 eV



smaller than that of  $\text{Na}_2\text{TeS}_3$ . Whereas the ionicity of the alkali cation and chalcogenide anion increases the band gaps, the structural differences between the *mP24* and *mC48* modifications have no significant influence on the optical properties. Almost equal band gaps are found for  $\text{Na}_2\text{TeSe}_3$ -*mP24* (1.87 eV) and  $\text{Na}_2\text{TeSe}_3$ -*mC48* (1.95 eV). The optical band gaps of  $\text{M}_2\text{TeQ}_3$  ( $\text{M} = \text{Li, Na}$ ;  $\text{Q} = \text{S, Se}$ ) are significantly higher than those of  $\text{Ag}_2\text{TeS}_3$  (0.35 eV), whereas the absorption behavior of  $\text{Na}_2\text{TeSe}_3$  and  $\text{Li}_2\text{TeS}_3$  is similar to that of ( $\text{Ag}_2\text{TeS}_3$ )  $\text{Rb}_2\text{S}_6$  and ( $\text{Ag}_2\text{TeS}_3$ )  $\text{Cs}_2\text{S}_6$  (2.0 eV).<sup>8</sup>

**Impedance Spectroscopy.** The temperature-dependent total conductivities of  $\text{Na}_2\text{TeS}_3$  and the isotypic modification of  $\text{Na}_2\text{TeSe}_3$ -*mP24* were analyzed by impedance spectroscopy. The Arrhenius plots of these compounds and the isotypic Li compounds in Figure 8 show a linear dependence of the logarithm of the specific conductivity  $\sigma_{\text{spez}}$  and the reciprocal temperature. The conductivities increase with increasing temperature.



**Figure 8.** Arrhenius plot of the total conductivities of isotypic  $\text{M}_2\text{TeQ}_3$  ( $\text{M} = \text{Li, Na}$ ;  $\text{Q} = \text{S, Se}$ ).

In the case of  $\text{Na}_2\text{TeS}_3$ , the conductivity ranges between  $1 \times 10^{-10} (\Omega \text{ cm})^{-1}$  at 370 K and  $4.1 \times 10^{-9} (\Omega \text{ cm})^{-1}$  at 470 K. Similar values are observed for  $\text{Na}_2\text{TeSe}_3$  [ $\sigma_{\text{spez}} = 8.7 \times 10^{-12} (\Omega \text{ cm})^{-1}$  at 370 K and  $\sigma_{\text{spez}} = 3.1 \times 10^{-9} (\Omega \text{ cm})^{-1}$  at 470 K]. The different ionicity of the chalcogens has only a very weak influence on the total conductivity. Larger differences were observed in the case of the lithium compounds. At 470 K,  $\text{Li}_2\text{TeS}_3$  has a lower conductivity [ $2.5 \times 10^{-8} (\Omega \text{ cm})^{-1}$ ] than  $\text{Li}_2\text{TeSe}_3$  [ $4.5 \times 10^{-6} (\Omega \text{ cm})^{-1}$ ].<sup>4</sup> Because the Nyquist plots of both sodium compounds do not show the typical linear spike at low frequencies, ionic contributions to the total conductivity can be regarded as quite small for  $\text{Na}_2\text{TeQ}_3$ . However, especially in the case of  $\text{Na}_2\text{TeS}_3$ , a certain ionic contribution to the total conductivity can be assumed, and therefore no clear trends can be expected for the activation energies from conductivity measurements. The activation energy for  $\text{Na}_2\text{TeS}_3$  (0.68 eV) was significantly smaller than that for  $\text{Na}_2\text{TeSe}_3$  (1.1 eV) and resembles the values for  $\text{Li}_2\text{TeS}_3$  (0.71 eV) and  $\text{Li}_2\text{TeSe}_3$  (0.68 eV).

**Thermal Analysis.**  $\text{Na}_2\text{TeS}_3$  showed its melting point at 914 K (onset temperature). In the case of  $\text{Na}_2\text{TeSe}_3$ , a heating cycle was measured for both modifications.  $\text{Na}_2\text{TeSe}_3$ -*mP24* melts at 748 K, whereas the melting point of  $\text{Na}_2\text{TeSe}_3$ -*mC48* is observed at slightly higher temperature (766 K). However, this difference can be regarded as affected by the heating rate; i.e., one should expect a phase transformation of the low-

temperature form to the high-temperature form and then an identical melting point.

## CONCLUSION

We present the crystal structure of  $\text{Na}_2\text{TeS}_3$ . It is the first reported compound in the ternary phase system  $\text{Na}-\text{Te}-\text{S}$ . It is isotypic with  $\text{Li}_2\text{TeQ}_3$  ( $\text{Q} = \text{S, Se}$ ). No phase transition was observed for  $\text{Na}_2\text{TeS}_3$ . This is contrary to  $\text{Na}_2\text{TeSe}_3$ . So far, only the *mC48* modification has been reported. Herein, we show the crystal structure of the second modification  $\text{Na}_2\text{TeSe}_3$ -*mP24*, which is also isotypic with  $\text{Li}_2\text{TeQ}_3$ . The formation can be controlled by the reaction temperature, heating time, and cooling rates. The lattice energies for both modifications differ only slightly in the case of  $\text{Na}_2\text{TeSe}_3$ . For  $\text{Na}_2\text{TeS}_3$ , the calculated lattice energy of a hypothetical  $\text{Na}_2\text{TeSe}_3$ -*mP24* is significantly higher than that for  $\text{Na}_2\text{TeS}_3$ -*mC48*.

The structural differences between the *mP24* and *mC48* modifications of  $\text{Na}_2\text{TeSe}_3$  hardly influence the optical band gap. They range between 1.87 eV ( $\text{Na}_2\text{TeSe}_3$ -*mP24*) and 1.95 eV ( $\text{Na}_2\text{TeSe}_3$ -*mC48*). On the other hand,  $\text{Na}_2\text{TeS}_3$  has a larger optical band gap (2.35 eV) because of the higher ionicity of sulfur. However, the total conductivities of  $\text{Na}_2\text{TeS}_3$  and  $\text{Na}_2\text{TeSe}_3$  are not influenced by the differences in their ionicity. They range from  $8.7 \times 10^{-12} (\Omega \text{ cm})^{-1}$  at 370 K and to  $3 \times 10^{-9} (\Omega \text{ cm})^{-1}$  at 470 K.

## ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.inorgchem.5b02105. Further details of the crystal structure investigations are available from Fachinformationszentrum (FIZ) Karlsruhe, 76344 Eggenstein-Leopoldshafen, Germany (fax + 49-7247-808-666; e-mail [crysdata@fiz-karlsruhe.de](mailto:crysdata@fiz-karlsruhe.de)) upon quoting the depository number.

Tables of the anisotropic displacement parameters  $U_{ij}$  (Tables S1, S3, and S5) of the selected interatomic distances and angles of the title compounds (Tables S2, S4, and S6), powder X-ray diffraction patterns (Figure S1), and plots of the electronic structures of  $\text{M}_2\text{TeQ}_3$  ( $\text{M} = \text{Li, Na}$ ;  $\text{Q} = \text{S, Se}$ ; Figure S2) (PDF)

CSD 430311 ( $\text{Na}_2\text{TeS}_3$ ) (CIF)

CSD 430313 ( $\text{Na}_2\text{TeSe}_3$ -*mP24*) (CIF)

CSD 430312 ( $\text{Na}_2\text{TeSe}_3$ -*mC48*) (CIF)

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### Notes

The authors declare no competing financial interest.

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