

Strong manipulation of the valley splitting upon twisting and gating in MoSe₂/CrI₃ and WSe₂/CrI₃ van der Waals heterostructures

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Van der Waals (vdW) heterostructures provide a rich playground to engineer electronic, spin, and optical properties of individual two-dimensional materials. We investigate the twist-angle and gate dependence of the proximity-induced exchange coupling in the monolayer transition-metal dichalcogenides (TMDCs) MoSe₂ and WSe₂ due to the vdW coupling to the ferromagnetic semiconductor CrI₃, from first-principles calculations. A model Hamiltonian, which captures the relevant band edges at the K/K' valleys of the proximitized TMDCs, is employed to quantify the proximity-induced exchange. Upon twisting from 0° to 30°, we find a transition of the TMDC valence band (VB) edge exchange splitting from about -2 to 2 meV, while the conduction band (CB) edge exchange splitting remains nearly unchanged at around -3 meV. For the VB of WSe₂ (MoSe₂) on CrI₃, the exchange coupling changes sign at around 8° (16°). We find that even at the angles with almost zero spin splittings of the VB, the real-space spin polarization profile of holes at the band edge is highly nonuniform, with alternating spin up-and spin-down orbitals. Furthermore, a giant tunability of the proximity-induced exchange coupling is provided by a transverse electric field of a few V/nm. Within our first-principles framework, we are limited to commensurate structures, and thus considered a maximum strain of 2.2% for the individual monolayers. By investigating different atomic stacking configurations of the strained supercells, we demonstrate that proximity exchange varies locally in space within experimentally realistic setups. We complement our *ab initio* results by calculating the excitonic valley splitting to provide experimentally verifiable optical signatures of the proximity exchange. Specifically, we predict that the valley splitting increases almost linearly as a function of the twist angle. Furthermore, the proximity exchange is highly tunable by gating, allowing to tailor the valley splitting in the range of 0 to 12 meV in WSe₂/CrI₃, which is equivalent to external magnetic fields of up to about 60 T. Our results highlight the important impact of the twist angle and gating when employing magnetic vdW heterostructures in experimental geometries.

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I. INTRODUCTION

Two-dimensional (2D) materials offer a huge variety of electronic, optical, spin, magnetic, and topological properties for future applications, which can be further tailored via external knobs [1–11]. For example, straining and gating of transition-metal dichalcogenides (TMDCs) can drive structural phase transitions, accompanied by significant modifications of the aforementioned properties [12–20]. Particularly, Song *et al.* [13] demonstrated a room temperature semiconductor-metal transition in thin film MoTe₂, due to the 2H-1T' structural phase transition via tensile strain of only 0.2%. Furthermore, if one wants to preserve the electronic properties of a 2D monolayer, and on-demand engineer its band structure, the concept of van der Waals (vdW) heterostructures and proximity effects come into play [5,21–23]. This concept is exemplified by graphene/TMDC heterostructures, in which the characteristic Dirac and semiconductor dispersions of the individual monolayers are well preserved, but the TMDC induces a giant (meV scale) proximity spin-orbit coupling (SOC) in graphene [24,25]. In terms of applications, graphene/TMDC heterostructures can be

employed for optospintronics [26,27] and spin-charge conversion [28–31], which is only possible due to the weakly coupled bilayer system, nicely demonstrating the power of vdW engineering.

Monolayer TMDCs—MoS₂, WS₂, MoSe₂, and WSe₂—are interesting on their own, since they are air-stable 2D semiconductors, which have a direct band gap in the optical range at the K/K' valleys of the hexagonal Brillouin zone [32–36]. External stimuli can be employed to engineer the electronic structure and the optical transitions [2,37]. For example, straining a monolayer WS₂ tunes the size of the band gap and leads to a direct-to-indirect band gap transition, as predicted by band structure calculations and observed in photoluminescence experiments [38–41]. Even though the optical excitations at K/K' valleys for unstrained monolayer TMDCs are energetically degenerate, they are helicity-sensitive—due to the lack of inversion symmetry combined with the presence of time-reversal symmetry—allowing to selectively address charge carriers at K or K' [42]. The valley is thus another potentially relevant parameter for information technology [7,43,44]. As experimentally demonstrated, valley polarizations of about 30% can be achieved in monolayer MoS₂ by optical pumping [45,46], enabling valleytronic devices [7,43,44,47,48].

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To achieve sizable valley polarizations, giant magnetic fields need to be applied. Indeed, a magnetic field of 1 Tesla yields only 0.1–0.2 meV of Zeeman splitting [49–56]. Proximity exchange fields perform much better and can induce several meVs, without significantly altering the band structure of TMDCs [57–69]. Of particular interest are all 2D vdW heterostructures, such as the recently considered $\text{WSe}_2/\text{CrI}_3$ bilayers [58,65,70–73], which show a few meV of proximity exchange. Remarkably, recent experiments have demonstrated all-optical magnetization switching in such heterostructures [73]. Moreover, an asymmetric magnetic proximity interaction has been recently observed in $\text{MoSe}_2/\text{CrBr}_3$ van der Waals heterostructures [74]. Nevertheless, most of these investigations have not considered the consequence of one important control parameter, which is the twist angle.

Only recently the relative twist angle between the monolayers has been recognized as a crucial degree of freedom. The most prominent examples are twisted bilayer and trilayer graphene, exhibiting magnetism and superconductivity due to strong correlations [75–99]. Other platforms for correlated physics are offered by twisted TMDCs [100–102] and twisted CrI_3 mono- and bilayers [103,104]. Regarding proximity effects, a recent study shows the sensitivity of the spin polarization, magnetic anisotropy, and Dzyaloshinskii-Moriya interaction on the twist angle in graphene/2H-VSeTe heterostructures [105]. Similarly, theoretical [106–112] and experimental [113] studies show that the strength of proximity SOC in graphene/TMDC and the proximity exchange in graphene/ $\text{Cr}_2\text{Ge}_2\text{Te}_6$ [114] heterostructures can be tuned by the twist angle.

In our previous work [72] on TMDC/ CrI_3 bilayers, we have only considered two twist angles, 0° and 30° , predicting a significant (meV scale) proximity exchange, with different sign for the TMDC valence band (VB) edge spin splitting at the two twist angles. No detailed study of the effect has been done for intermediate twist angles to see if the sign change is systematic or accidental. Another important question relates to the proximity exchange of the conduction band (CB): does the direction of the proximity exchange remain independent of the twist angle, as suggested by the 0° and 30° predictions? What is the variation of the magnitude of the proximity exchange as a function of the twist angle? Could there be a scenario such as predicted for graphene/ $\text{Cr}_2\text{Ge}_2\text{Te}_6$ [114] that the effective proximity exchange becomes antiferromagnetic at an intermediate angle? What are the real-space spin-polarization profiles of the conduction and valence electrons in the TMDCs for different twist angles? In the present study, we systematically address these questions.

Recent band structure investigations of $\text{WSe}_2/\text{CrI}_3$ heterostructures by Ge *et al.* [115] revealed a relative valley splitting enhancement of about 1000% by twisting, in good agreement with Ref. [72]. They have shown that mainly the TMDC VB splitting is responsible for the massive twist tunability of the valley splitting. In addition they also found a reversal of the VB splitting and related valley splitting upon twisting, but did not analyze this fact. Overall, the tunability of the valley splitting is attributed to the magnetic proximity effect and related to the enhancement of the Cr magnetic moments upon twisting. However, Ge *et al.* [115] considered just four twist angles and rather large strains of up to 5.5%,

with respect to the lattice constants determined from density functional theory (DFT), which significantly alters both monolayer dispersions and CrI_3 magnetic properties [39,116]. In addition, an electric field dependence was performed only for one specific twist angle of 16.1° .

Here, we systematically investigate the twist-angle and electric-field dependence of the proximity-induced exchange coupling in $\text{MoSe}_2/\text{CrI}_3$ and $\text{WSe}_2/\text{CrI}_3$ bilayers, from first principles. We consider seven different angles between 0° and 30° with a maximum strain—needed for commensurability of the unit cells—of about 2.2% applied to the monolayers when employing experimentally determined lattice constants. Within the heterostructures the TMDCs preserve their characteristic dispersion, while the magnetic insulator substrate provides proximity exchange, splitting the relevant TMDC VB and CB edges (in the absence of SOC, see below) at K/K' valleys. Depending on the relative twist angle the band edge splittings can be markedly different, and even the spin ordering can be reversed. In particular, when twisting from 0° to 30° , the TMDC CB edge splitting remains nearly unchanged at around -3 meV. In contrast, there is a smooth transition of the TMDC VB edge spin splitting from about -2 to 2 meV (the spin ordering changes). The finding about the reversal of the spin ordering is consistent with Ref. [72], but here we additionally find that it happens in an almost linear fashion upon twisting, and we identify the twist angles at which the transition happens: about 8° (16°) for WSe_2 (MoSe_2). The origin of this reversal is traced back to the twist-angle dependent backfolding of the TMDC K/K' valleys into the CrI_3 Brillouin zone. By investigating different atomic stacking configurations of the strained supercells, we show that proximity exchange varies locally in space within experimentally realistic heterostructures.

Furthermore, we reveal a rather high tunability of the proximity-induced exchange coupling by applying a transverse electric field of a few V/nm across all the twisted $\text{WSe}_2/\text{CrI}_3$ heterostructures. In Ref. [72], the electric field tunability was investigated for the 0° structure only. We also introduce a minimal model Hamiltonian to describe the twist- and gate-tunable proximity effects in the TMDCs due to CrI_3 . The Hamiltonian together with the fitted parameters provide an effective description for the TMDC band edge physics at K/K' valleys, which is relevant for calculating magneto-optical Kerr effect [117,118], absorption spectra [72,119,120], or coupled spin Hall and valley Hall effects [42]. Finally, we give specific predictions for experimentally verifiable optical signatures of the proximity exchange effects, by calculating the excitonic absorption spectra employing the Bethe-Salpeter equation. In addition to 0° and 30° angles [72], we provide the full twist-angle and electric-field dependence of the valley splitting. We find a rather high tunability of the valley splitting of the first intralayer exciton peak, ranging from 0 to 12 meV by gating and twisting in $\text{WSe}_2/\text{CrI}_3$ bilayers, equivalent to external magnetic fields of up to about 60 T.

The paper is organized as follows. In Sec. II, we first address the structural setup and summarize the calculation details for obtaining the electronic structures of the twisted TMDC/ CrI_3 bilayers. In Sec. III, we introduce the model Hamiltonian that captures the relevant band edges at K/K' valleys of the proximitized TMDCs, which are used to fit the

TABLE I. Structural information for the TMDC/CrI₃ heterostructures. We list the twist angle ϑ between the layers, the number of atoms (NoA) in the heterostructure supercell, the number n_k for the k -point sampling, the lattice constants and biaxial strains ε applied to the TMDCs and CrI₃, the calculated dipoles and the relaxed interlayer distances d_{int} of the MoSe₂ (WSe₂) structures.

ϑ (deg)	NoA	n_k	a_{TMDC} (Å)	$\varepsilon_{\text{TMDC}}$ (%)	a_{CrI_3} (Å)	$\varepsilon_{\text{CrI}_3}$ (%)	dipole (debye)	d_{int} (Å)
0.0000	20	30	3.3608	2.21	6.7215	-2.12	0.0507 (0.1082)	3.5467 (3.5000)
8.2132	243	9	3.3400	1.58	6.7472	-1.75	0.0322 (0.5475)	3.5888 (3.5661)
10.1583	149	12	3.2880	0	6.9189	0.76	0.3590 (0.6993)	3.5858 (3.5749)
16.1021	63	24	3.2880	0	6.8444	-0.33	0.0881 (0.2474)	3.5930 (3.5490)
21.7868	140	12	3.3610	2.22	6.7189	-2.16	-0.0725 (0.2432)	3.5666 (3.5499)
25.2850	183	9	3.3350	1.43	6.7604	-1.55	0.0033 (0.4402)	3.5885 (3.5627)
30.0000	243	9	3.3400	1.58	6.7492	-1.72	-0.0347 (0.5360)	3.5837 (3.5640)

first-principles results. In Sec. IV, we show and discuss exemplary calculated electronic structures, along with the model Hamiltonian fits. The twist-angle and gate dependence of the proximity-induced exchange couplings is also addressed. In Sec. V, we show and discuss the twist-angle and electric-field tunability of the valley splitting, as calculated from the absorption spectra employing the Bethe-Salpeter equation. Finally, in Sec. VI, we conclude the manuscript.

II. GEOMETRY SETUP & COMPUTATIONAL DETAILS

The twisted MoSe₂/CrI₃ and WSe₂/CrI₃ heterostructures for the DFT calculations are set up with the ATOMIC SIMULATION ENVIRONMENT (ASE) [121] and the CELLMATCH code [122], implementing the coincidence lattice method [123,124]. Monolayers of TMDCs and CrI₃ are based on hexagonal unit cells, with lattice constants of $a = 3.288$ Å (MoSe₂), $a = 3.282$ Å (WSe₂), and $a = 6.867$ Å (CrI₃) [125–127], which need to be strained in the twisted heterostructures, in order to form commensurate supercells for periodic DFT calculations. Since MoSe₂ and WSe₂ have nearly the same lattice constant, we set them as equal in the following. In Table I, we summarize the main structural information for the twist angles we consider. All our heterostructures have C₃ symmetry. In Fig. S1 [128], we show top views of all employed supercell geometries. In total, we investigate seven different angles between 0° and 30°. Especially these angles are suitable for DFT calculations, since strain applied to the monolayers is at maximum 2.2%. We already know that biaxial strain strongly influences the band gap, spin-orbit splittings and spin-valley properties of monolayer TMDCs [39,56], as well as the band gap of CrI₃ [116], therefore we keep the strain as small as possible. In addition, the number of atoms is kept below 250. Otherwise, also other angles could be investigated, but beyond reasonable strain limits and above a computationally feasible number of atoms in the structure.

The electronic structure calculations and structural relaxation of the TMDC/CrI₃ heterostructures are performed by DFT [129] with QUANTUM ESPRESSO [130,131]. Self-consistent calculations are carried out with a k -point sampling of $n_k \times n_k \times 1$. The number n_k is listed in Table I for all twist angles and depends on the number of atoms in the heterostructure. In addition, n_k is limited by our computational power. Nevertheless, for large supercells the heterostructure Brillouin

zone is small and only a few k points are necessary to get converged results.

We perform open shell calculations that provide the spin-polarized ground state of the CrI₃ monolayer. A Hubbard parameter of $U = 3.0$ eV is used for Cr d orbitals [72,132]. We use an energy cutoff for charge density of 520 Ry and the kinetic energy cutoff for wave functions is 65 Ry for the scalar relativistic pseudopotentials with the projector augmented wave method [133] with the Perdew-Burke-Ernzerhof exchange correlation functional [134]. SOC is neglected, since we are mainly interested in the twist-angle dependence of the proximity-induced exchange coupling. Moreover, we have already demonstrated that SOC and proximity exchange are rather independent in the investigated bilayers [72]. For the relaxation of the heterostructures, we add DFT-D2 vdW corrections [135–137] and use quasi-Newton algorithm based on trust radius procedure. Dipole corrections [138] are also included to get correct band offsets and internal electric fields. In order to simulate quasi-2D systems, we add a vacuum of about 24 Å to avoid interactions between periodic images in our slab geometry. For proper interlayer distances, we allow the atoms of the TMDCs, as well as the Cr atoms of the CrI₃, to relax their z coordinates, while the I atoms are allowed to move freely, because they form a distorted octahedral surrounding around the Cr atoms [139]. Relaxation is performed until every component of each force is reduced below 10^{-3} [Ry/ a_0], where a_0 is the Bohr radius.

After relaxation, we calculate the mean interlayer distances, d_{int} , from the z coordinates of interfacial Se and I atoms. The interlayer distances are nearly independent of the twist angle, about 3.55 Å, and are listed in Table I. In Fig. 1, we show the general structural setup of our TMDC/CrI₃ heterostructures, where the TMDC resides above CrI₃, with the magnetization \mathbf{M} along z direction specifying the spin quantization axis (spin up = z , spin down = $-z$). When we apply the transverse electric field (modeled by a zigzag potential), a positive field also points along z direction.

III. MODEL HAMILTONIAN

We want to describe proximity exchange effects in the monolayer TMDCs that are due to the magnetic insulator substrate CrI₃. Similar to our recent work [72], we employ a minimal model Hamiltonian to describe the band edges of the

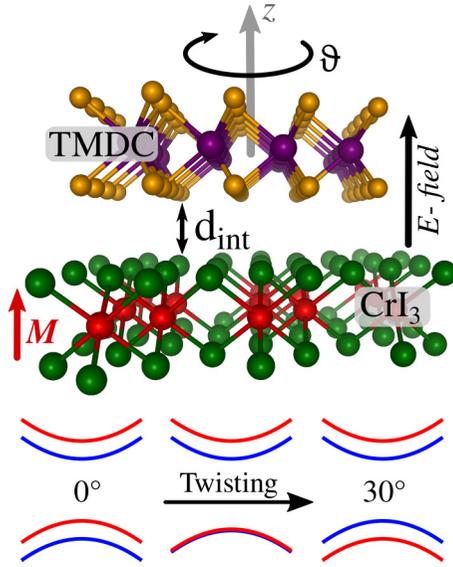


FIG. 1. 3D view of a TMDC (MoSe₂ or WSe₂) on CrI₃, where we define the interlayer distance, d_{int} . We twist the TMDC by an angle ϑ around the z axis, with respect to the magnetic semiconductor CrI₃, with magnetization M along the z direction. The twist-angle evolution of the proximitized TMDC band edges are sketched. Red bands are polarized spin up, while blue bands are polarized spin down. By twisting from 0° to 30° , the TMDC VB edge splitting, which is due to proximity exchange, first vanishes and then reverses sign.

TMDC close to K and K' valleys, in the presence of proximity exchange

$$\mathcal{H} = \mathcal{H}_0 + \mathcal{H}_\Delta + \mathcal{H}_{\text{ex}}, \quad (1)$$

$$\mathcal{H}_0 = \hbar v_F s_0 \otimes (\tau \sigma_x k_x + \sigma_y k_y), \quad (2)$$

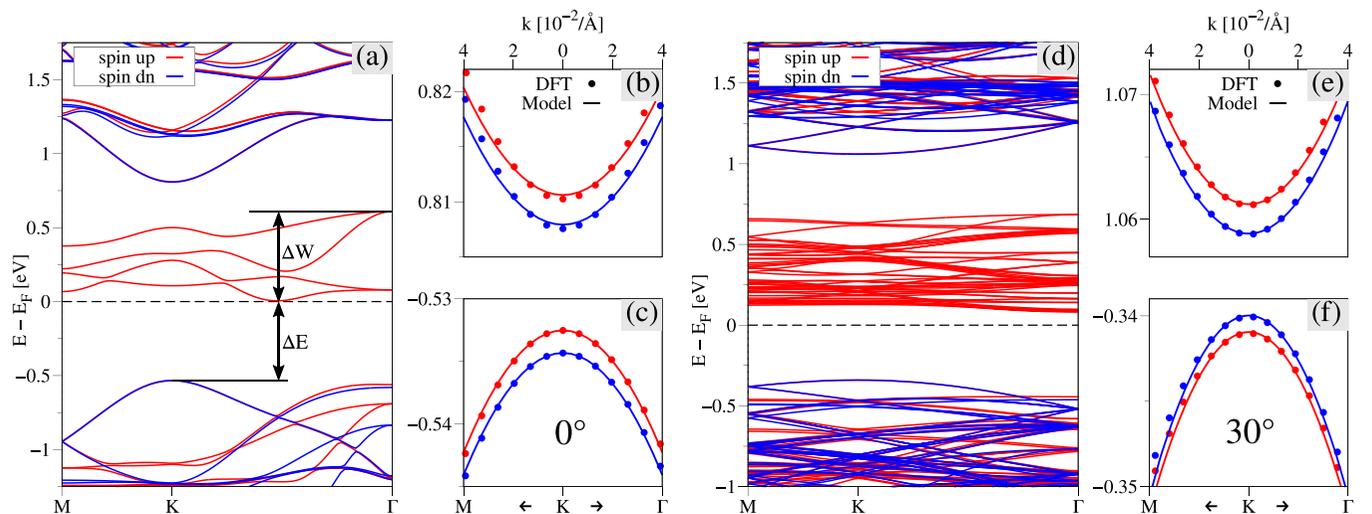


FIG. 2. (a) DFT-calculated band structure of the MoSe₂/CrI₃ heterostructure for a twist angle of 0° along the high-symmetry path M - K - Γ . Bands in red are spin up, while bands in blue are spin down. We define the heterostructure band gap ΔE , between the TMDC VB edge at K , and the minimum of spin polarized CrI₃ in-gap states. We also define the bandwidth ΔW of the in-gap states. (b) Zoom to the MoSe₂ CB edge near the K point, showing proximity exchange split bands. Symbols are DFT data and solid lines are the fitted model Hamiltonian results. (c) Same as (b), but for the VB edge. [(d)–(f)] Same as [(a)–(c)], but for a twist angle of 30° . Comparing (c) and (f), the spin ordering of bands is reversed.

$$\mathcal{H}_\Delta = \frac{\Delta}{2} s_0 \otimes \sigma_z, \quad (3)$$

$$\mathcal{H}_{\text{ex}} = -s_z \otimes (B_c \sigma_+ + B_v \sigma_-). \quad (4)$$

The valley index is $\tau = \pm 1$ for K/K' point and v_F is the Fermi velocity. The Cartesian components k_x and k_y of the electron wave vector are measured from K/K' . The pseudospin Pauli matrices are σ_i acting on the [conduction band (CB), valence band (VB)] subspace and spin Pauli matrices are s_i acting on the (\uparrow, \downarrow) subspace, with $i = \{0, x, y, z\}$. The parameter Δ denotes the orbital gap of the spectrum. For short notation, we introduce $\sigma_\pm = \frac{1}{2}(\sigma_0 \pm \sigma_z)$. With a magnetic substrate layer, proximity exchange coupling will split the TMDC CB edge by $2B_c$ and the VB edge by $2B_v$. The four basis states we use are $|\Psi_{\text{CB}}, \uparrow\rangle$, $|\Psi_{\text{VB}}^\tau, \uparrow\rangle$, $|\Psi_{\text{CB}}, \downarrow\rangle$, and $|\Psi_{\text{VB}}^\tau, \downarrow\rangle$. The wave functions are $|\Psi_{\text{CB}}\rangle = |d_{z^2}\rangle$ and $|\Psi_{\text{VB}}^\tau\rangle = \frac{1}{\sqrt{2}}(|d_{x^2-y^2}\rangle + i\tau|d_{xy}\rangle)$, corresponding to CB and VB at K/K' [32].

IV. TWIST- AND GATE-TUNABLE PROXIMITY EXCHANGE

A. Twist angle dependence

In Fig. 2(a), we show the DFT-calculated band structure of the MoSe₂/CrI₃ heterostructure for a twist angle of 0° . The band edges of the TMDC can be nicely recognized at the K point at about -0.5 and 0.8 eV with respect to the Fermi level. The states ranging from about 0 to 0.5 eV are spin polarized in-gap states that originate from the CrI₃ layer. Therefore the heterostructure forms a type-II band alignment with a bandgap ΔE of about 0.5 eV. In our heterostructures, small charge transfer is present between the monolayers, leading to a finite intrinsic electric (dipole) field. The dipole of the heterostructure, which controls the band alignment and ΔE , depends on the twist angle, as summarized in Table II. However, we

TABLE II. Fit parameters of Hamiltonian \mathcal{H} for the TMDC/CrI₃ heterostructures for different twist angles. We summarize the Fermi velocity v_F , the orbital gap Δ , proximity exchange parameters B_c and B_v , the heterostructure band gap ΔE , and the CrI₃ in-gap states bandwidth ΔW , as defined in Fig. 2(a).

	ϑ (deg)	v_F ($\times 10^5$ m/s)	Δ (eV)	B_c (meV)	B_v (meV)	ΔE (eV)	ΔW (eV)
MoSe ₂	0.0000	4.358	1.343	-1.343	-0.901	0.538	0.603
	8.2132	4.631	1.403	-1.070	-0.262	0.432	0.601
	10.1583	4.626	1.555	-1.427	-0.365	0.330	0.485
	16.1021	4.622	1.556	-1.332	-0.012	0.337	0.535
	21.7868	4.575	1.340	-1.253	0.213	0.463	0.617
	25.2850	4.048	1.415	-1.221	0.415	0.417	0.587
	30.0000	4.622	1.401	-1.176	0.484	0.422	0.603
WSe ₂	0.0000	5.873	1.403	-1.460	-0.918	0.377	0.600
	8.2132	5.954	1.479	-1.175	-0.020	0.255	0.601
	10.1583	6.093	1.674	-1.553	0.099	0.165	0.484
	16.1021	6.084	1.674	-1.500	0.705	0.178	0.535
	21.7868	5.839	1.403	-1.318	0.894	0.291	0.615
	25.2850	5.216	1.495	-1.290	1.415	0.248	0.587
	30.0000	5.940	1.477	-1.226	1.525	0.251	0.604

believe that the strain that is applied to the monolayers has the most significant impact on the band alignment, as previous studies of graphene-based heterostructures suggest [108,114]. Therefore the intrinsic band alignment that should be present in experiments can be read off from the 16.1° structure, with marginally strained monolayers (0.33% of strain). In addition, the CrI₃ in-gap states bandwidth ΔW depends on the lattice constant used in the heterostructure. Compressive (tensile) strain increases (decreases) ΔW . Finally, also the TMDC band gap Δ depends on the lattice constant and is strain tunable [39]. However, the 0° and 30° structures have rather similar strain, and still the proximity exchange is opposite for the VB; this strongly suggests that the variation of the proximity exchange with the twist angle is not much affected by the modest strain we have in the supercells.

In Figs. 2(b) and 2(c), we show the zooms to the MoSe₂ band edges near the K point for the 0° structure. The bands experience proximity-induced exchange splitting of about 2 meV. The spin ordering is the same for CB and VB, with the spin down band lower in energy. The fit parameters from Table II nicely reproduce the TMDC band edges. In Figs. 2(d)–2(f), we summarize the results for the MoSe₂/CrI₃ heterostructure for a twist angle of 30°. Overall, the band structure features remain the same, but with many more bands due to the larger supercell size compared to the 0° structure. Looking at the relevant TMDC band edges, proximity exchange splitting is still present, but the VB spin ordering is *reversed* compared to the 0° case. The CB proximity exchange remains the same. Below and in Ref. [128], we further elaborate on the origin of this reversal.

In Fig. 3, we summarize the twist-angle dependence of the proximity exchange parameters, B_c and B_v , as listed in Table II for the MoSe₂ and WSe₂ heterostructures. We find that the twist angle barely influences the CB proximity exchange parameter B_c , being fixed at around -1.2 meV. In contrast, the VB proximity exchange parameter, B_v , is negative for 0°, *vanishes and reverses sign* at about 16° (8°) for MoSe₂ (WSe₂), and is positive for a twist angle of 30°. The atomic arrangement (stacking) of the monolayers, for a fixed twist

angle, can also influence proximity exchange [71,72], see Ref. [128] for more details. Apparently, for the different twist angles, we have chosen the stackings such that B_c shows a kink around 8°.

For completeness, in Ref. [128], we investigate different atomic stacking configurations and also briefly address the twist angles between 30° to 60°. Overall, we find that in large supercells the stacking only marginally influences proximity exchange couplings, while in small supercells—such as the 0° structure with only 20 atoms—the VB edge splitting can be tuned in sign and magnitude by the atomic arrangement, see also Fig. 4. We believe that an averaging effect takes place in larger supercell geometries, where a lot of atoms are involved in forming the TMDC band edge states, which can locally pick up different exchange fields, but globally lead to the same band edge splitting. In other words, experimentally the proximity exchange will vary locally in space across the heterostructure.

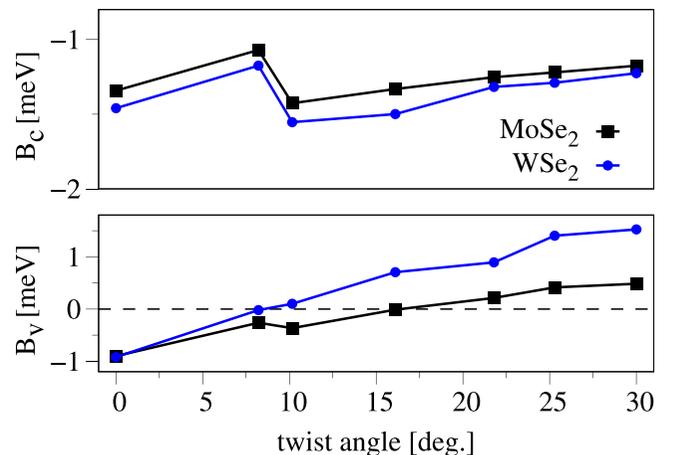


FIG. 3. Calculated twist-angle dependence of the proximity exchange couplings B_c and B_v for the MoSe₂/CrI₃ and WSe₂/CrI₃ heterostructures.

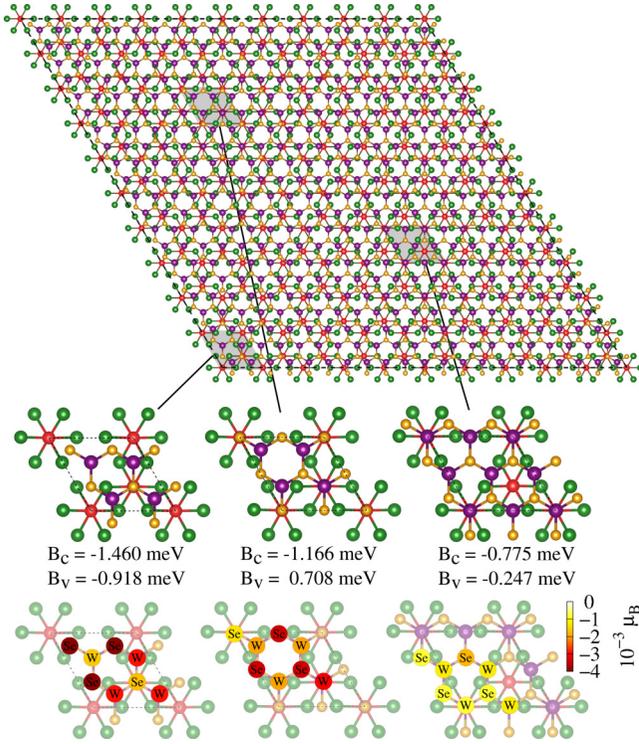


FIG. 4. (Top) An experimentally realistic 0° $\text{WSe}_2/\text{CrI}_3$ heterostructure (2123 atoms), where we consider a 21×21 supercell of WSe_2 on a 10×10 supercell of CrI_3 , corresponding to strains of 0% and 0.5%. (Middle) Different high-symmetry stackings of the 0° strained heterostructure are locally indicated, with their corresponding proximity exchange parameters B_c and B_v . (Bottom) Calculated proximity-induced local magnetic moments (colored spheres) of the interfacial Se and W atoms for the different stackings.

Such a local variation can be seen in Fig. 4 (Fig. S2), where we also show the proximity-induced local magnetic moments of interfacial Se and W atoms of different 0° stacking configurations (of all investigated heterostructures). We find that the induced magnetic moments depend sensitively on the stacking and correlate with the proximity exchange parameters B_c and B_v . With high-resolution vector magnetometry and gradiometry, based on nitrogen-vacancy centers in diamond [140–142], one can potentially resolve the demonstrated local variations of proximity exchange.

In Figs. 4 and S19, we explicitly map different high-symmetry stackings of our strained 0° structure to an experimentally more realistic setup, further confirming the local variation of proximity exchange. In fact, the proximity exchange parameter B_c varies by a factor of 2 among the investigated stackings, while the parameter B_v can be even reversed in sign and magnitude. However, one has to relate this with experimental limitations. Even though there can be local variations of proximity exchange, typical photoluminescence techniques are not capable of addressing only a specific stacking. Then, even for the 0° case, an ensemble of stackings will be probed, thus leading to an average proximity exchange. Based on the findings for different stackings, the average exchange splittings follow the same predictions as we provide in Fig. 3 (VB and CB splittings negative for 0°). Moreover,

twisting from 30° to 60° leads to similar proximity exchange couplings—apart from differences in magnitude owed to the atomic stacking—as twisting from 30° to 0° .

As mentioned above, the strain is a byproduct of our structural setup, leading to nontrivial influences on band gaps and magnetic properties of our monolayers [39,116]. In Ref. [128], we also investigate the influence of the strain distribution for the $\text{WSe}_2/\text{CrI}_3$ structure with 0° twist angle. We find that the exchange parameters can be changed by at maximum $\pm 20\%$ by tuning the strain, but their signs and the order of magnitude remains. In other words, the presented results are representative for experimental conditions (no or weak strain). Based on the findings for 0° , one can expect a similar strain dependence for the other investigated twist angles, where strain is similar or smaller compared to 0° , see Table I. Especially for the 10.2° and 16.1° structures, where strain is small, we believe our calculations provide a close correspondence to experimental results.

Regarding experiments, one can expect similar results for few layer CrI_3 , including even and odd number of layers. In Ref. [72], we have shown that the proximity exchange coupling is restricted to interfacial layers, i.e., mainly the topmost CrI_3 layer influences the TMDC. In particular, we have shown that proximity exchange for TMDC on bilayer CrI_3 in ferro- and antiferromagnetic configuration is essentially unchanged. Of course, the sign of the proximity effect does depend on the magnetization direction of the interfacial CrI_3 layer, but not the magnitude.

B. Electric field tunability

In experimental geometries, gating is a useful tool to further control and tailor the proximity exchange coupling in TMDC/magnetic-semiconductor heterostructures. In the following, we consider the experimentally relevant $\text{WSe}_2/\text{CrI}_3$ heterostructure [58,65,70] and perform a gate scan, in the range of ± 2 V/nm, for all twist angles. The electric-field and twist-angle dependence of the proximity exchange parameters B_c and B_v is summarized in Fig. 5, while the full fit results are summarized in Table S2 [128]. For all angles and electric fields we consider, the CB proximity exchange parameter, B_c , stays negative and within the limits of about -0.5 to -2.5 meV. In contrast, the VB proximity exchange parameter B_v can be even stronger controlled by the electric field and the twist angle. As already mentioned, we find a crossover from positive to negative values at around 8° twist angle for WSe_2 . Furthermore, the applied field allows to tune B_v in a wide range of values. Considering all angles, the tunability is giant and within the limits of about -1.5 to 4 meV.

C. Reversal of the valence band proximity exchange splitting

The reversal of the TMDC VB edge splitting upon twisting is one of the most interesting findings, see Fig. 3. A similar reversal of the proximity-induced exchange coupling has been recently found in graphene/ $\text{Cr}_2\text{Ge}_2\text{Te}_6$ bilayers when twisting from 0° to 30° [114]. The reversal was attributed to the different coupling of C orbitals to the spin up and spin down band manifolds of the magnetic substrate for the different twist angles. What is the mechanism for the investigated TMDC/ CrI_3

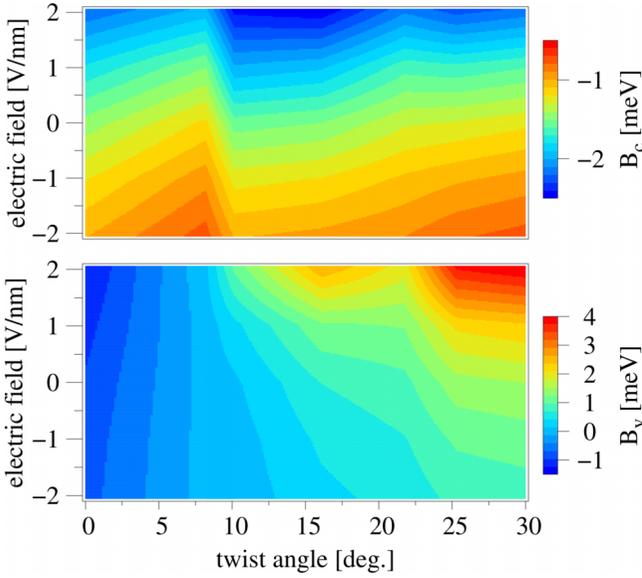


FIG. 5. Calculated electric-field and twist-angle dependence of the proximity-induced exchange parameters B_c (top) and B_v (bottom) for the $\text{WSe}_2/\text{CrI}_3$ heterostructure, interpolated from Table S2.

bilayers? For this purpose, we analyze the twisted $\text{WSe}_2/\text{CrI}_3$ heterostructures in more detail. In particular, we look at the three most relevant twist angles, namely, 0° , 8.2° , and 30° , corresponding to the cases when the WSe_2 VB edge proximity exchange splittings are negative, almost zero, and positive.

A similar analysis should hold for MoSe_2 , but the splitting vanishes at a different twist angle. In the following, we want to address the main points, while a more detailed discussion is given in Ref. [128].

To gain information about the proximity-induced exchange coupling, we look at the TMDC band edge spin polarizations in real space. Considering only TMDC CB edge states near the K point, the calculated spin polarization is negative, independent of the twist angle and in agreement with the band edge dispersion, predominantly formed by $\text{W } d_{z^2}$ orbitals, and localized only on the WSe_2 layer (see Fig. 6). In other words, the low-energy TMDC CB edge states, which are spin-split due to proximity exchange, do not contain a spin polarization from the CrI_3 layer. Indeed, mainly in the spin down channel the high-energy TMDC and CrI_3 bands are coupled, as one can see in the projected band structures for the three twist angles (see Figs. S3, S7, and S10). Considering second-order perturbation theory, the dominant coupling between spin down high-energy levels can repel the TMDC spin down band edge states to lower energies compared to the spin up ones, consistent with the observed splittings in Figs. 2(b) and 2(e). Additionally considering the coupling to CrI_3 spin up in-gap states leads to the same conclusion.

The explanation for the VB edge is a bit more involved. In Fig. 6, we also show the calculated TMDC VB edge spin polarizations for 0° , 8.2° , and 30° . In the case of 0° , the VB edge splitting is negative [the dispersion is sketched in Fig. 6(a)]. This leads to the positive spin polarization we can see in Fig. 6(a), which is spread uniformly across the TMDC and

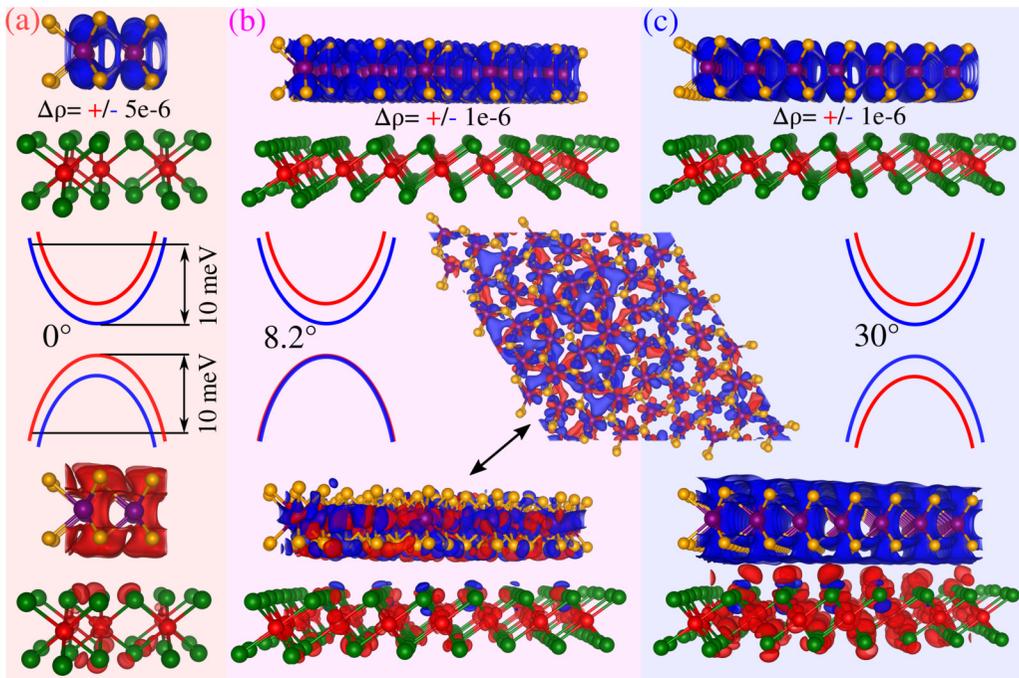


FIG. 6. Calculated real-space TMDC band edge spin polarizations, $\Delta\rho = \rho_\uparrow - \rho_\downarrow$, for the twisted $\text{WSe}_2/\text{CrI}_3$ heterostructures. The background color is used to group the subfigures for the three twist angles (a) 0° , (b) 8.2° , and (c) 30° . (a) (Middle) Sketch of the proximitized TMDC band edges for 0° , where we indicate the energy windows from which we calculate the spin densities $\rho_{\uparrow/\downarrow}$. (Top/bottom) Calculated band edge spin polarization, taking into account only CB/VB states. The color red (blue) corresponds to $\Delta\rho > 0$ ($\Delta\rho < 0$). The isosurfaces correspond to isovalues (units \AA^{-3}) as indicated. (b) The same as (a), but for 8.2° . For the VB, we also show a top view, with removed CrI_3 layer, to show the nonuniform spin polarization on WSe_2 . (c) The same as (a), but for 30° .

predominantly formed by W $d_{xy} + d_{x^2-y^2}$ and Se p orbitals. In addition, Cr $d_{zx} + d_{zy}$ and I $p_x + p_y$ orbitals contribute to the TMDC VB edge spin polarization. For the 30° case, we find the opposite and a uniform negative band edge spin polarization on the TMDC, see Fig. 6(c), again in agreement with the dispersion. For 8.2° , the VB edge splitting almost vanishes and we find a highly nonuniform spin polarization, see Fig. 6(b), with a multipole character around the W atoms. On average, the spin polarization is zero and therefore the band splitting is absent, but this does not rule out the complex spin structure we find. In fact, since the proximity exchange can sensitively depend on the stacking, as we have shown especially for the 0° supercell, one can expect local variations of the band edge spin polarizations also in experimentally realistic setups of the other twist angles. Analyzing the TMDC VB states in more detail, there is apparently a delicate balance between the coupling to CrI₃ spin polarized in-gap and VB states. What is different is the backfolding of the TMDC K point into the Brillouin zone of CrI₃ for different twist angles. For example, for 0° the TMDC K point folds back near the CrI₃ K point, while at 30° it folds back near the Γ point, see Fig. S5. Since the interlayer coupling is different at different k points, and also different CrI₃ band manifolds and orbitals play a role, this could indeed lead to the reversal of the TMDC VB edge splitting.

V. TWIST- AND GATE-TUNABLE VALLEY SPLITTING

In order to provide insight into the optical signatures of the proximity exchange, we evaluate the valley splitting, i.e., the energy separation between the absorption peaks with opposite circular polarization (σ^+ and σ^-). Combining the proximity exchange parameters, B_c and B_v , from Table S2 [128], with the intrinsic SOC parameters for MoSe₂ and WSe₂ monolayers (see, for example, Ref. [39]), we can provide valuable insights regarding the twist-angle and electric-field dependence of the valley splitting. Besides the Hamiltonian terms given in Eq. (1), the SOC term reads

$$H_{\text{soc}} = \tau s_z \otimes (\lambda_c \sigma_+ + \lambda_v \sigma_-), \quad (5)$$

with parameters $\lambda_v = 94.56$ meV and $\lambda_c = -9.647$ meV for MoSe₂, and $\lambda_v = 241.79$ meV and $\lambda_c = 13.9$ meV for WSe₂, taken from Ref. [72].

To incorporate excitonic effects and calculate the absorption spectra, we apply the robust formalism of the effective Bethe-Salpether equation with the electron-hole Coulomb interaction mediated by the Rytova-Keldysh potential, similar to Refs. [72,120]. From the calculated absorption, we can extract the valley splitting, i.e., the energy difference between the first A exciton peaks at K and K' (with opposite circular polarization). For the excitonic calculations, the BSE is solved numerically using 101×101 k points in a square region with sides ranging from -0.5 to 0.5 \AA^{-1} , thus leading to a k -point spacing of $\Delta k = 10^{-2} \text{ \AA}^{-1}$ along each direction. Additionally, the Coulomb potential is averaged in a submesh of 101×101 points covering an area of Δk^2 around each k point. The screening length used in the calculations is $r_0 = 51$ (45) \AA for MoSe₂ (WSe₂), from Ref. [143], and the dielectric constant for CrI₃ is taken as $\epsilon = 1.8$, from Ref. [144].

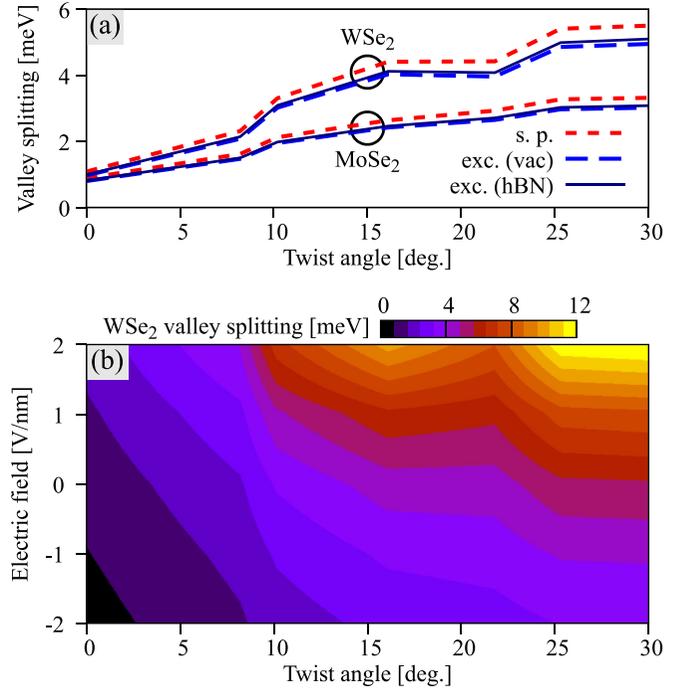


FIG. 7. (a) Valley splitting for the MoSe₂/CrI₃ and WSe₂/CrI₃ bilayers calculated from the single-particle (abbreviated by s.p., short dashed lines), and from the excitonic absorption (abbreviated as exc.) considering the region above the TMDC to be vacuum (long dashed lines) or hexagonal boron nitride (solid lines). The single-particle values follow nicely the excitonic calculations. (b) Valley splitting for the WSe₂/CrI₃ bilayer as function of the twist angle (x axis) and applied electric field (y axis) using the single-particle values, calculated via Eq. (6).

In the single-particle limit, i.e., where no excitonic effects are taken into account, the valley splitting can be obtained analytically by solving the Hamiltonians (1) and (5) and is given by

$$\begin{aligned} \text{VS} &= (E_{c\uparrow,K} - E_{v\uparrow,K}) - (E_{c\downarrow,-K} - E_{v\downarrow,-K}) \\ &= -2(B_c - B_v). \end{aligned} \quad (6)$$

In Fig. 7(a), we reveal the behavior of the valley splitting as a function of the twist angle for MoSe₂/CrI₃ and WSe₂/CrI₃. For both cases, the valley splitting increases as the twist angle gets larger, almost in a linear fashion. The role of excitonic effects due to different environments on top of the TMDC is also explored: air/vacuum with $\epsilon = 1$ (long dashed lines) and hexagonal boron nitride with $\epsilon = 4.5$ [145–147] (solid lines). The single-particle values (short dashed lines), obtained from Eq. (6), are also shown and follow closely the trends of the excitonic valley splitting extracted from the absorption spectra (shown in the Supplemental Material [128]), suggesting that it is possible to grasp valuable insight into the valley splitting just by knowing the exchange parameters B_c and B_v extracted from DFT calculations. Our results reveal that the twist-angle is a crucial parameter for the assembly of TMDC/ferromagnetic-semiconductor vdW heterostructures and, particularly, it has a sizable impact on the magnitude of the valley splitting.

To highlight the extraordinary tunability of the valley splitting by the combined effect of the twist angle and gating, we show in Fig. 7(b) a complete color map for the $\text{WSe}_2/\text{CrI}_3$ system, using the single-particle valley splitting expression of Eq. (6). On the bottom left corner of the figure, the valley splitting nearly vanishes for 0° twist angle and ~ -2 V/nm, whereas on the top right corner, the valley splitting reaches very large values of approximately 12 meV for 30° and ~ 2 V/nm. For a fixed twist angle, the dependence of the valley splitting is almost linear, in agreement with our previous calculations [72] and experimentally demonstrated in $\text{MoSe}_2/\text{CrBr}_3$ samples [68].

VI. SUMMARY

We have shown that, by employing first-principles calculations on large supercells, one can tailor the proximity exchange coupling in TMDC/ CrI_3 bilayers by twisting and gating. The previously observed reversal of the proximity exchange is demonstrated to result from a systematic, essentially linear dependence of the exchange in the VB on the twist angle. We do not find any significant variation in the CB.

Analyzing the real-space spin-resolved electronic density for the band edges, we find that the CB electrons exhibit a uniform spatial spin polarization for all the investigated twist angles. In contrast, the hole spin polarization undergoes remarkable transformations as going from 0° to 30° . While at the two limit angles the spin polarizations are highly uniform, albeit opposite (corresponding to the opposite spin splittings), at the twist angles where the spin splitting reverses sign the spin polarization is highly nonuniform, alternating between

spin up and spin down orbitals, arising from a subtle balance between orbital- and spin-resolved hybridization of the TMDC and CrI_3 orbitals. By comparing different stacking configurations, we have also demonstrated that the induced magnetic moments, proximity exchange, and the spin polarizations can vary locally in space within experimentally realistic setups.

Further substantial tunability of the band edge splittings is provided by a transverse electric field of a few V/nm. A low-energy model Hamiltonian, with fitted parameters that nicely reproduce the relevant DFT-calculated band edges of the proximitized TMDCs, has been employed to reveal the experimentally verifiable signatures of the CrI_3 exchange field on the valley splitting of the first TMDC exciton peak. Particularly, we predict that the valley splitting increases linearly with respect to the twist angle. Furthermore, in $\text{WSe}_2/\text{CrI}_3$ heterostructures the valley splitting shows an unprecedented high tunability upon twisting and gating, ranging from 0 to 12 meV, equivalent to external magnetic fields of up to 60 T. Our results also stress the importance of documenting the twist angle when employing magnetic vdW heterostructures in experiments.

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